

Nb_3Sn for SRF applications

High efficiency cavities for
future accelerators



Grigory Ereameev

Friday, August 9, 2019

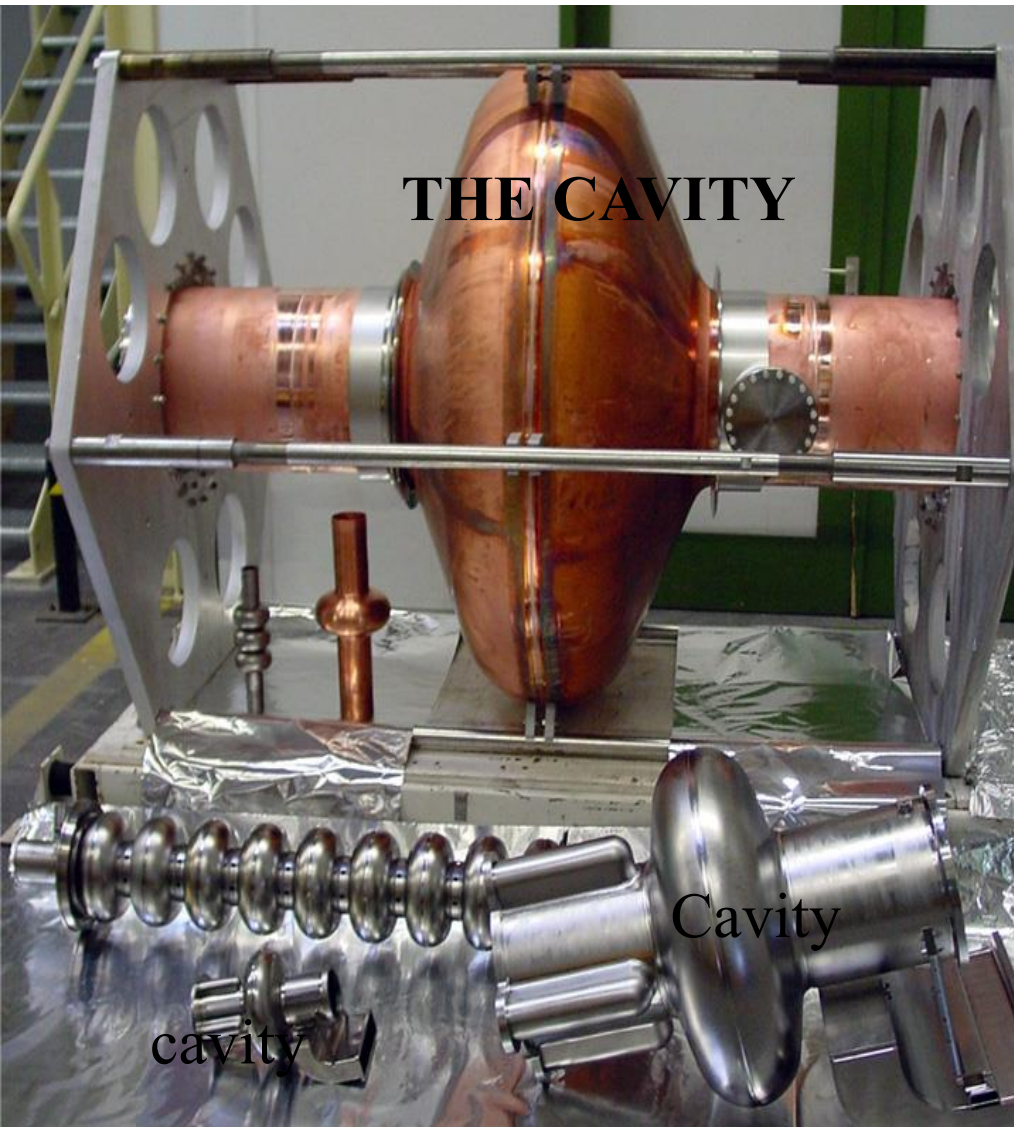
Acknowledgements

- Michael Kelley, Uttar Pudasaini, Jay Tuggle
- Hani Elsayed-Ali, Md. Nizam Sayeed, Jean Delayen, Jayendrika Tiskumara
- Gigi Ciovati, Charlie Reece, Bob Rimmer, Anne-Marie Valente-Feliciano, Larry Phillips, Peter Kneisel, John Mammosser
- ✓ N. Hasan, C. Mounts, K. Macha, W. Oren, A. Solopova, M. Wright, M. Drury, J. Grames, R. Kazimi, W. Crahen, M. Poelker, T. Powers, J. Preble, R. Suleiman, Y. Wang, M. Wright, A. Hutton, H. Areti et al.
- ✓ Matthias Liepe, Ryan Porter, Sam Posen, et al.
- ✓ JLab technical staff

Outline

- Motivation
- Background
- Current status
- Path forward

Niobium and its limitations



Niobium – best superconducting properties among all pure metals:

- $T_c \sim 9.25$ K;
- $H_c \sim 2000$ Oe;
- $R_{bcs} \sim .00001$ m Ω at 2 K

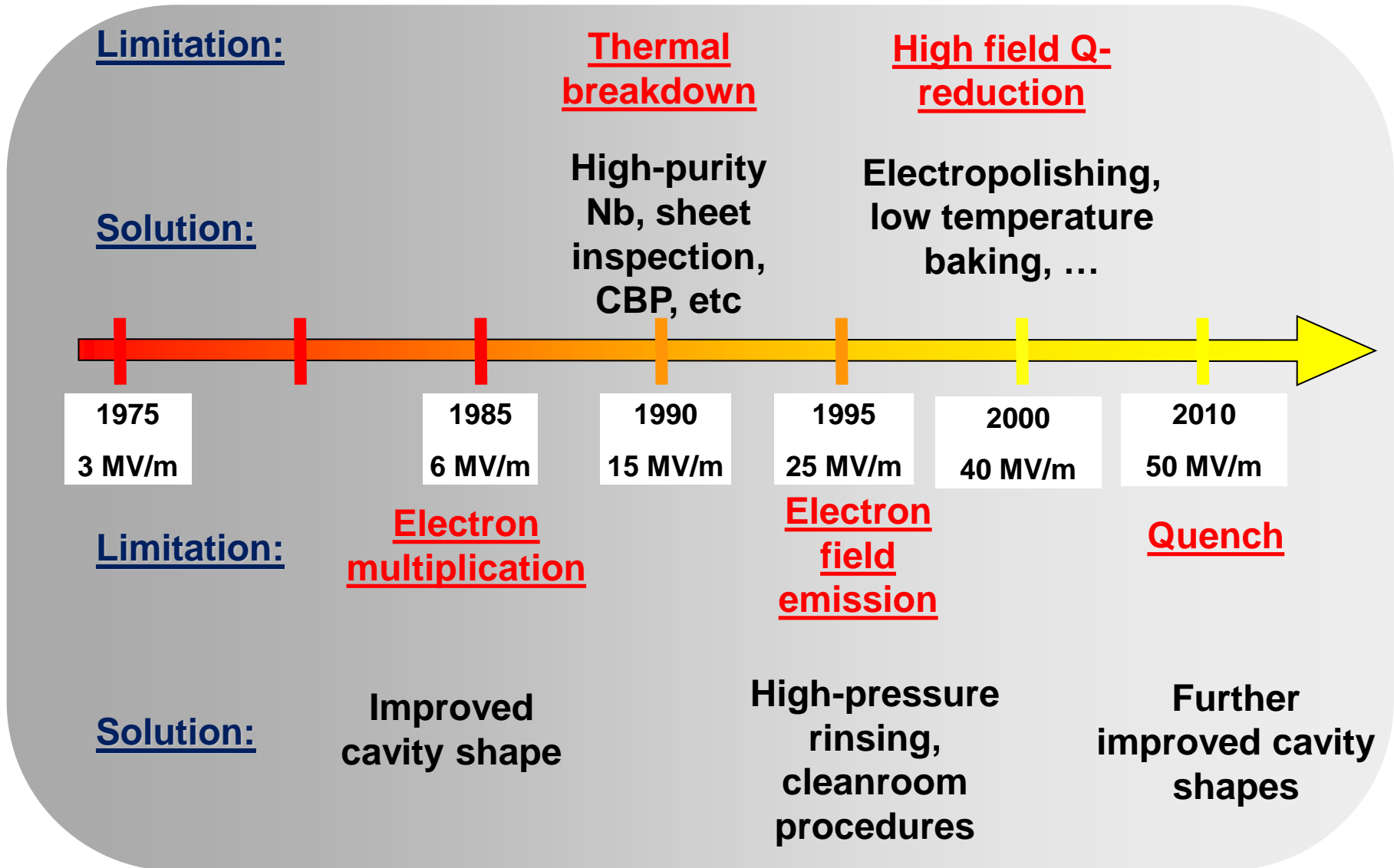
Nb : $R_s \sim .00001$ m Ω

Cu: $R_s \sim 10$ m Ω

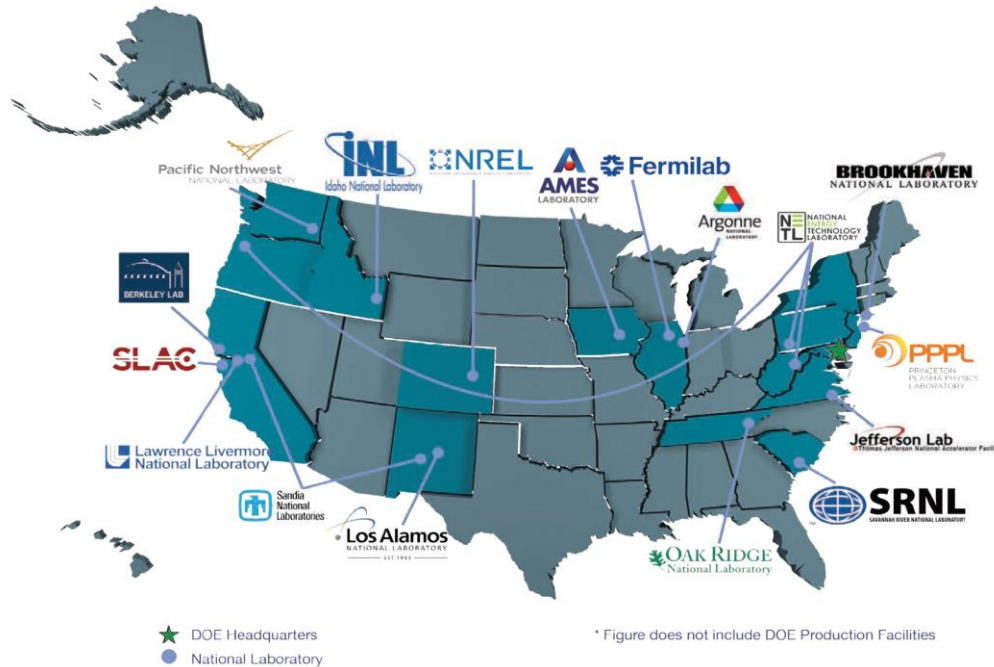


$Q^{nb} \sim 10^{11}$ up to
 $E_{acc} = 50 \cdot 10^6$ Volts
per meter

Niobium and its limitations

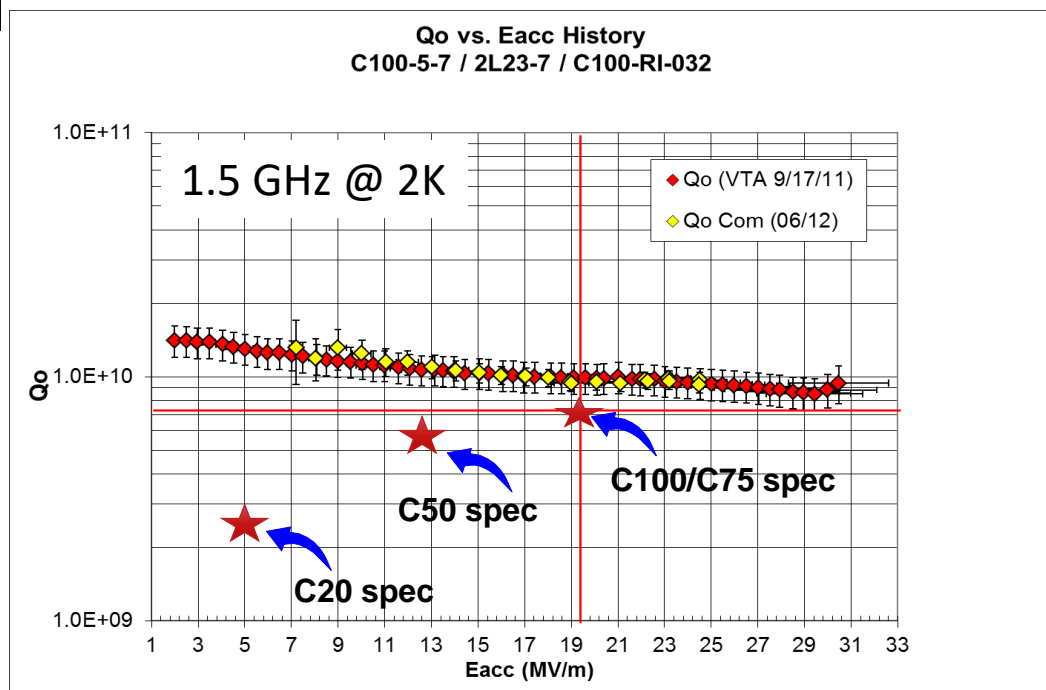
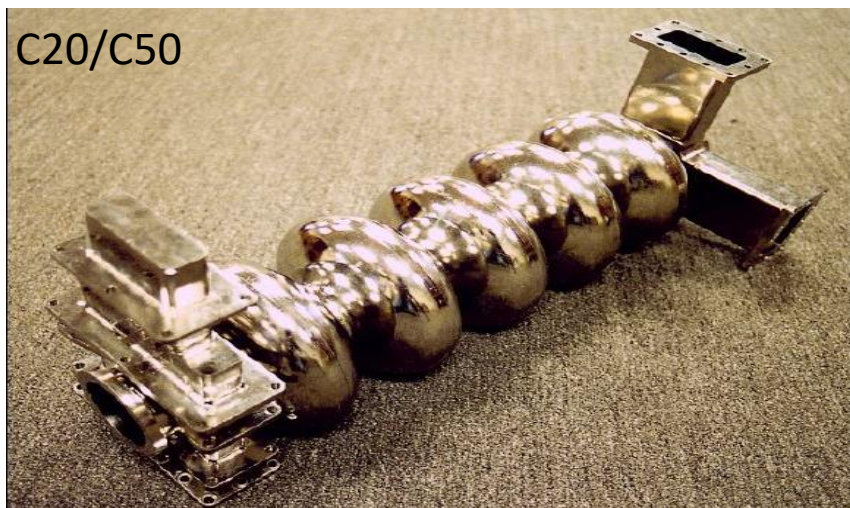


Jefferson Lab Overview

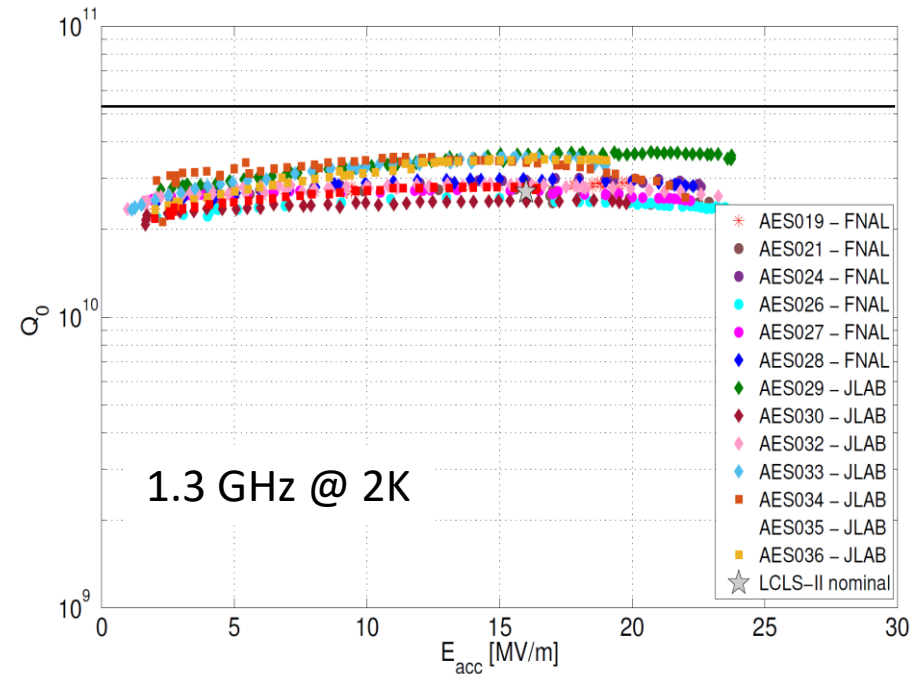
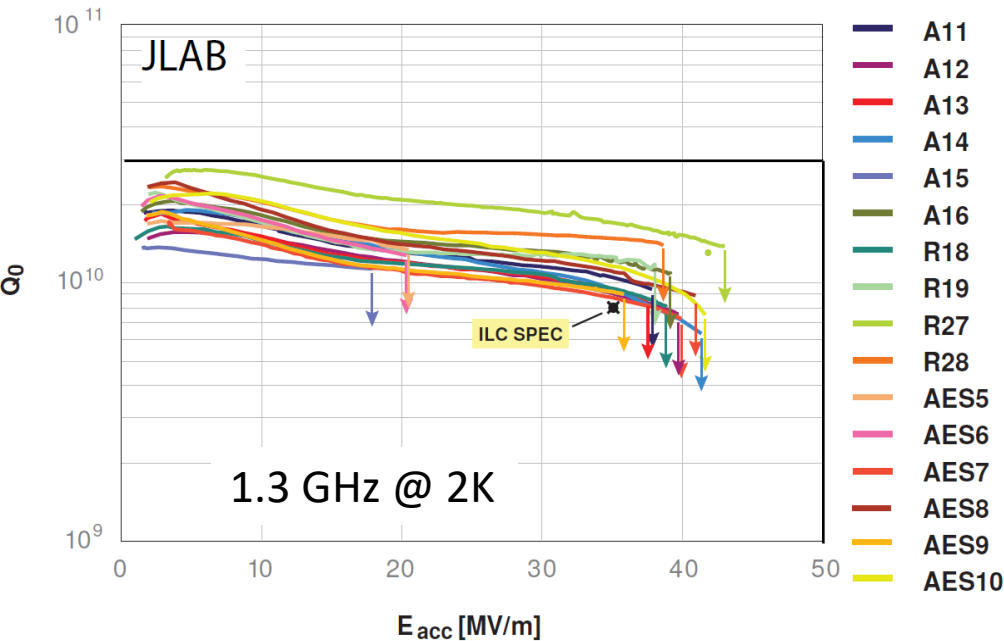


- **Core Competencies**
 - Accelerator Science and Technology
 - Large Scale User Facilities/Advanced Instrumentation
 - Nuclear Physics
- **Mission Unique Facilities**
 - Continuous Electron Beam Accelerator Facility

CEBAF SRF cavities



JLab SRF is a part of global efforts to improve SRF technology



THE INTERNATIONAL LINEAR COLLIDER
Technical Design Report | Volume 3.i: Accelerator R&D

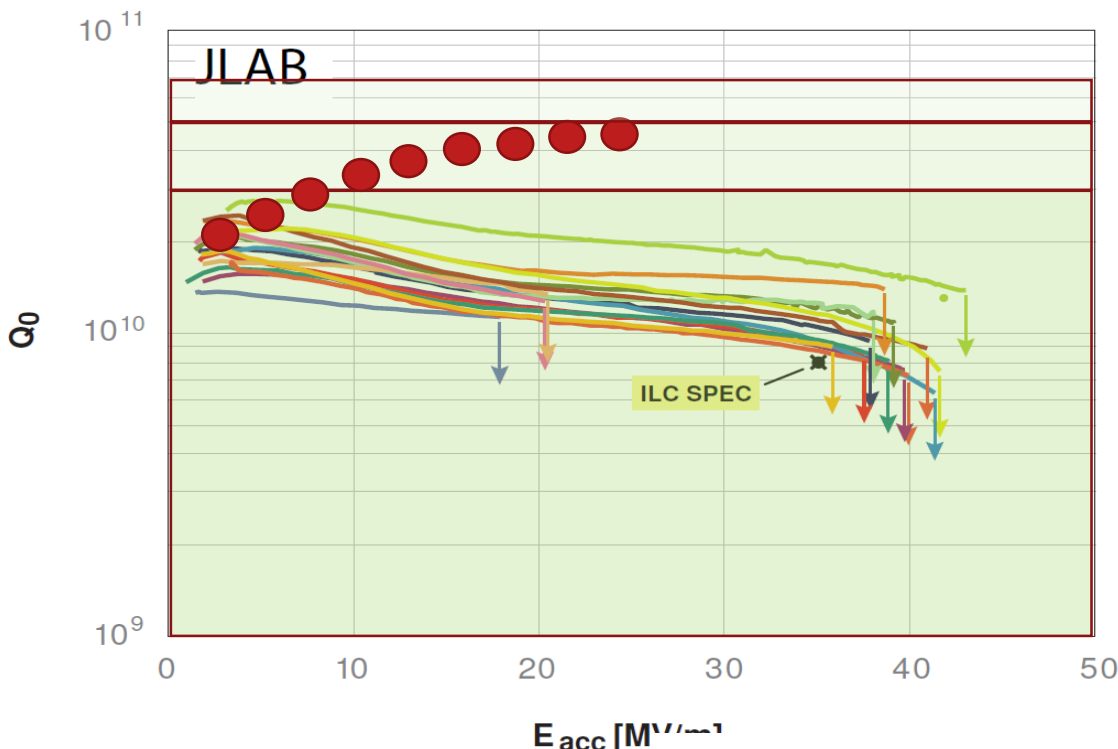
**LCLS-II SRF CAVITY PROCESSING PROTOCOL DEVELOPMENT AND
BASELINE CAVITY PERFORMANCE DEMONSTRATION, MOPB033,
SRF'15**

$$R_{BCS} \cong \frac{R_n}{\sqrt{2}} \left(\frac{\hbar \omega}{\pi \Delta} \right)^2 \frac{\sigma_1}{\sigma_n} \cong A \sqrt{\rho_n} e^{-\frac{\Delta}{K_B T}}$$

$$\Delta = 1.45 \text{ meV} \Rightarrow R_s \geq 5 \text{ n}\Omega @ 2\text{K} @ \sim 1 \text{ GHz}$$

$$\underline{H_c \sim 200 \text{ mT} \Rightarrow H_{sh} \sim 240 \text{ mT} \Rightarrow E_{acc} \sim 50 \text{ MV/m}}$$

Ideas for the future



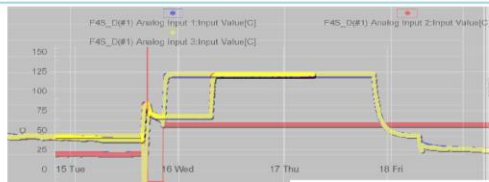
THE NEXT SRF TECHNOLOGIES

KEK / SOKENDAI
TAKAYUKI KUBO
<http://researchmap.jp/kubotaka/>
LCWS2016
2016.12.05

The Graduate University
for Advanced Studies

AES9
AES10

So, what caused the improvement?



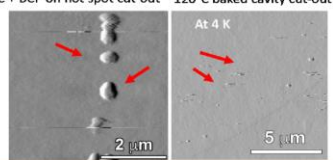
The α and β Peaks in Cold-Worked Niobium

M. W. STANLEY*, Z. C. SZKOPAK
Department of Metallurgy and Materials Technology, University of Surrey,
Guildford, GU2 7XH

Internal-friction measurements (at 1 c/sec) have been carried out over the temperature range from 40 to 300°K, on stainless specimens deformed at room temperature. In the as-cold-worked material, a broad peak at about 110°K is usually observed. This peak decreases with the amount of deformation and decreases with increasing interstitial impurity content. On subsequent annealing, the height of the peak decreases by about 50% over the temperature range from 80 to 180°K, and to negligible values from 180 to 300°K.

As a result of annealing for 2 h at 10°K, a group of peaks (β peak) occurred at about 300°K. The β peaks are independent of the amount of deformation prior to annealing and the interstitial impurity content. On further annealing, the relative strength of the peaks increased with temperature up to about 180°K, reached a maximum between 180 and 200°K, and subsequently gradually decreases to negligible low values at about 300°K.

The α peak, and its variation with deformation, impurity content, and annealing, can be accounted for in terms of relaxation mechanisms involving dislocations (i.e. a Bordoni or Frank-type process observed in f.c.c. metals). This is a generally accepted concept at present. The β peaks, on the other hand, could only be adequately accounted for by relaxation processes involving complexes of deformation-induced point defects and interstitial impurities.



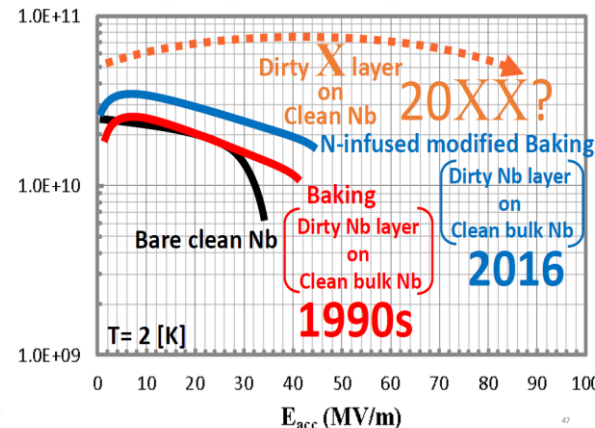
Nb₃Sn

I suggest another key mechanism is at play

- In addition to surface barrier (superheating) there is a "time barrier"
 - There should be enough time for vortices to nucleate/dissipate
- Vortex nucleation is governed by the characteristic time scale of order parameter changes, so-called τ_Δ
 - If flux penetration/dissipation is happening or not depends on the relation between τ_Δ and RF period T_{rf}
 - $\tau_\Delta > T_{rf} \Rightarrow$ vortex-induced dissipation is delayed beyond Hsh
 - $\tau_\Delta < T_{rf} \Rightarrow$ Hc1 and superheating become more relevant – more DC-like
 - $\tau_\Delta \gg T_{rf} \Rightarrow$ vortices don't matter as they never form
- $\tau_\Delta \sim \tau_{GL} \ll 1$ ns is only relevant for gapless superconductors (which Nb is not) > was understood by e.g. Tinkham and Bezuglii in late 1980s
- For gapped superconductors at low T: $\tau_\Delta \sim \tau_E \gg 1$ ns for Nb

Present situation

will be changed by advanced layered structures



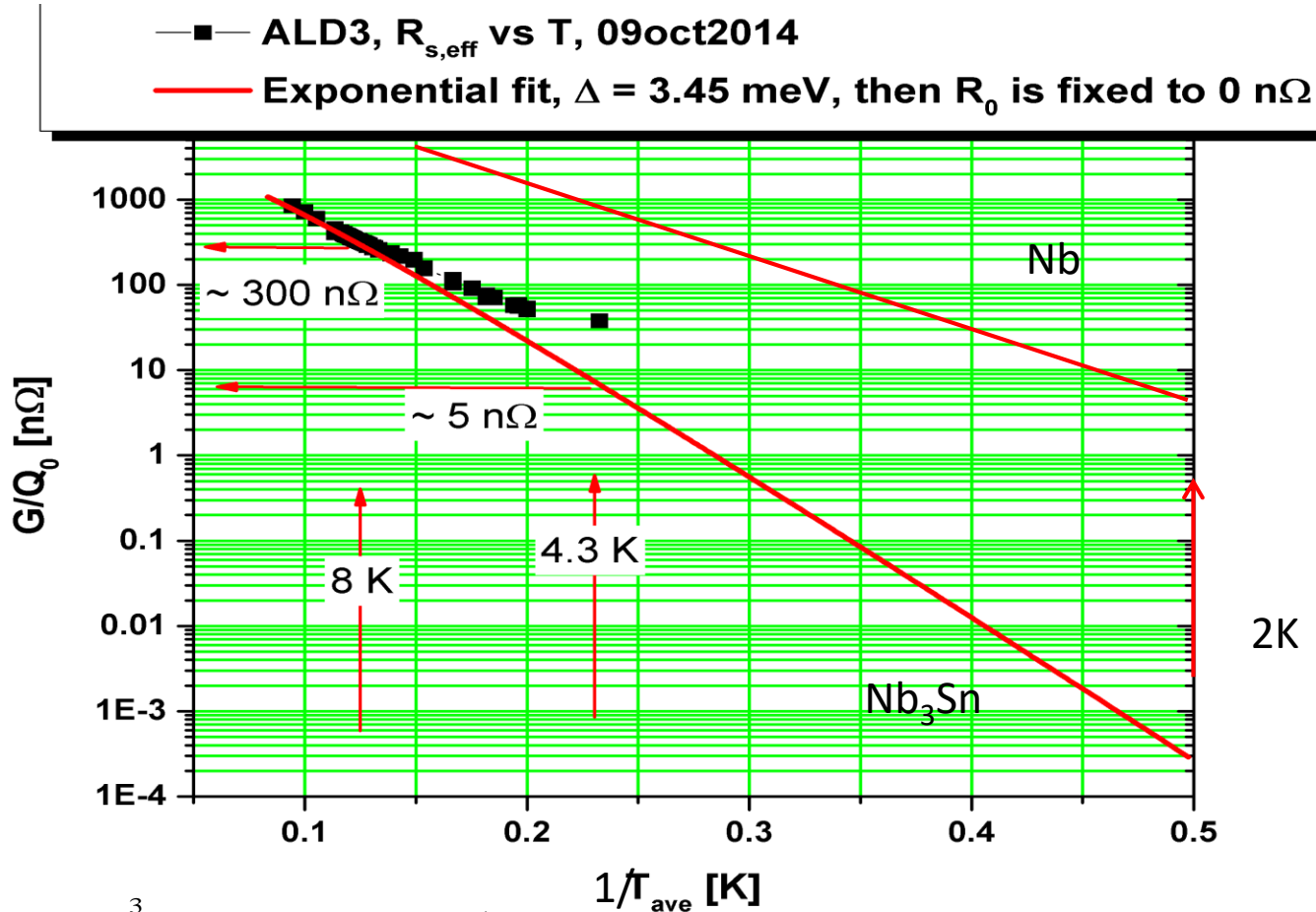
What are the better SRF Materials?

T_c [K]	ρ_n [$\mu\Omega\text{cm}$]	Δ [meV]	$H_c(0)$ [T]	$H_{sh}(0)$ [T]	$H_{c1}(0)$ [T]	$\lambda(0)$ [nm]	Material
9.25	0.1	1.45	~ 0.2	~ 0.24	~ 0.17	40	Nb
17.2	70	2.6	~ 0.23	~ 0.19	~ 0.02	~ 200	NbN
17.5	35	3.0	~ 0.28	~ 0.24	~ 0.03	~ 151	NbTiN
18.3	5	3.1	~ 0.54	~ 0.45	~ 0.05	~ 85	Nb ₃ Sn
40	2	2.3/7.1	~ 0.43	~ 0.27	~ 0.03	~ 140	MgB ₂

- s-wave superconductor
- large energy gap
- high H_{sh}
- low normal-conducting resistivity

Material	Nb	Nb ₃ Sn
T_c [K]	9.25	18.3
ρ_n [$\mu\Omega\text{cm}$]	0.1	~ 5
$H_{sh}(0)$ [T]	0.24	~ 0.45
Δ [meV]	1.45	~ 3.1
Q^{BCS} @ 2K	$\sim 5 \cdot 10^{10}$	$\sim 5 \cdot 10^{14}$
Q^{BCS} @ 4K	$\sim 5 \cdot 10^8$	$\sim 5 \cdot 10^{10}$
E_{acc} [MV/m]	~ 50	~ 100

Nb3Sn properties and perspectives



$$R_{BCS} \cong \frac{R_n}{\sqrt{2}} \left(\frac{\hbar \omega}{\pi \Delta} \right)^{\frac{3}{2}} \frac{\sigma_1}{\sigma_n} \cong A \sqrt{\rho_n} e^{-\frac{\Delta}{k_B T}}$$

$$\Delta \sim 3.1 \text{ meV} \Rightarrow \underline{R_s \geq 5 \text{ n}\Omega @ 4.3\text{K} @ \sim 1 \text{ GHz}}$$

$$\underline{H_c \sim 540 \text{ mT} \Rightarrow H_{sh} = 0.84 \cdot H_c \sim 450 \text{ mT} \Rightarrow \underline{E_{acc} \sim 100 \text{ MV/m}}}$$

Nb₃Sn: past and present

Nb₃Sn for SRF!! ... not exactly new

- 1953, discovered by B. Matthias et al.
- 1962, Saur and Wurm
- 1973, Siemens AG
- 1974, Karlsruhe
- 1974, Cornell University
- 1975, University of Wuppertal
- 1986, CERN
- ...
- ...
- ...
- ...
- 2009, Cornell University
- 2012, Jefferson Lab
- 2015, Fermilab

High T_c?

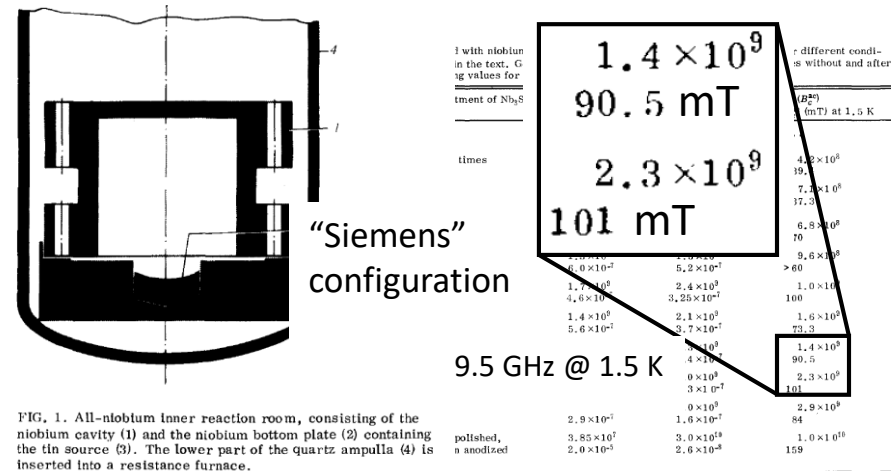
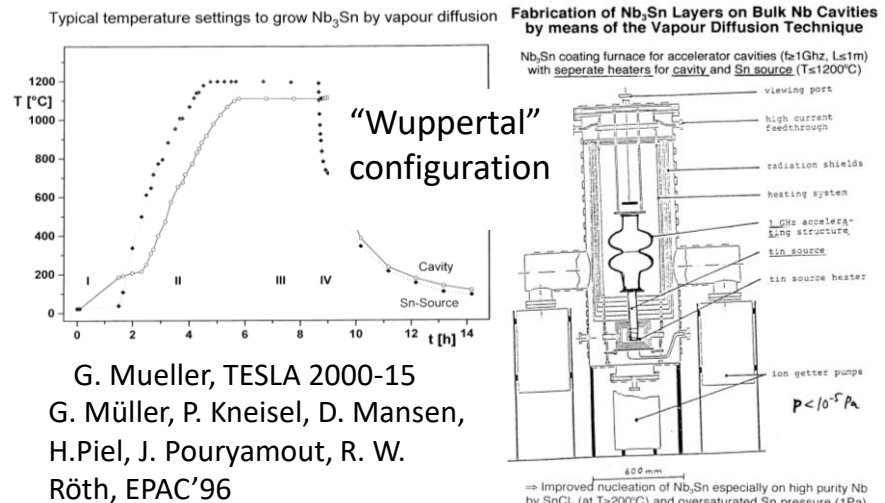
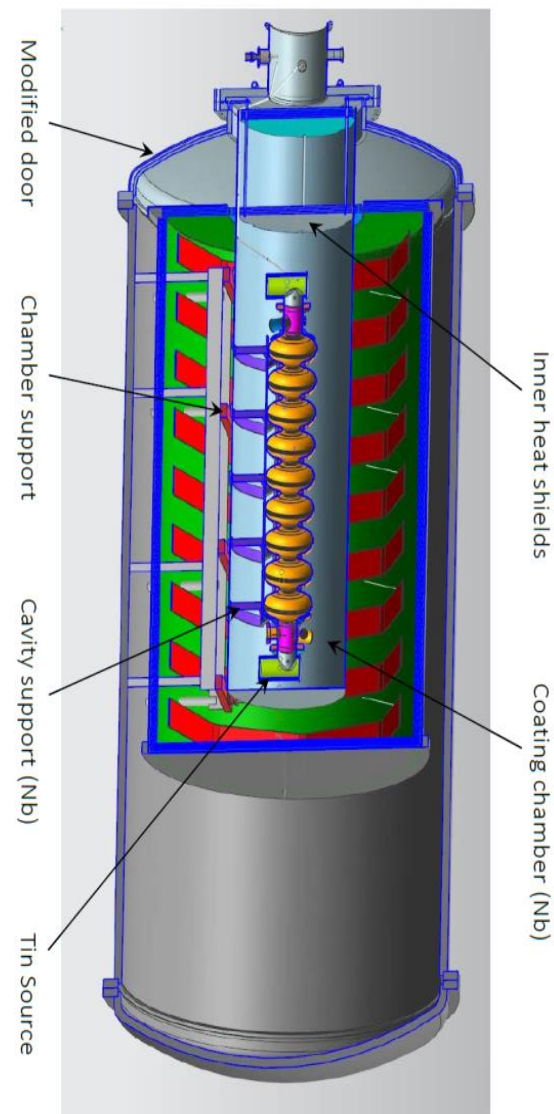
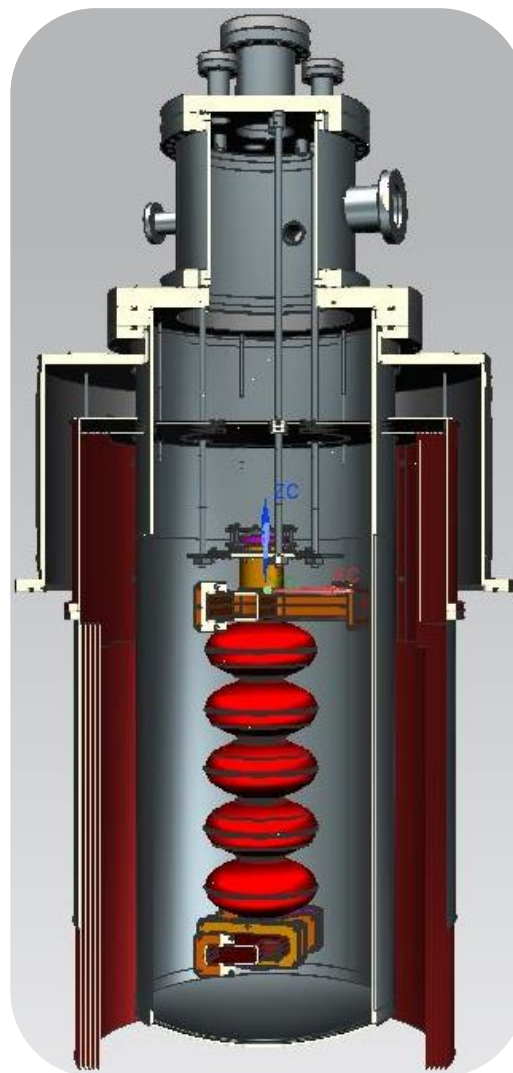
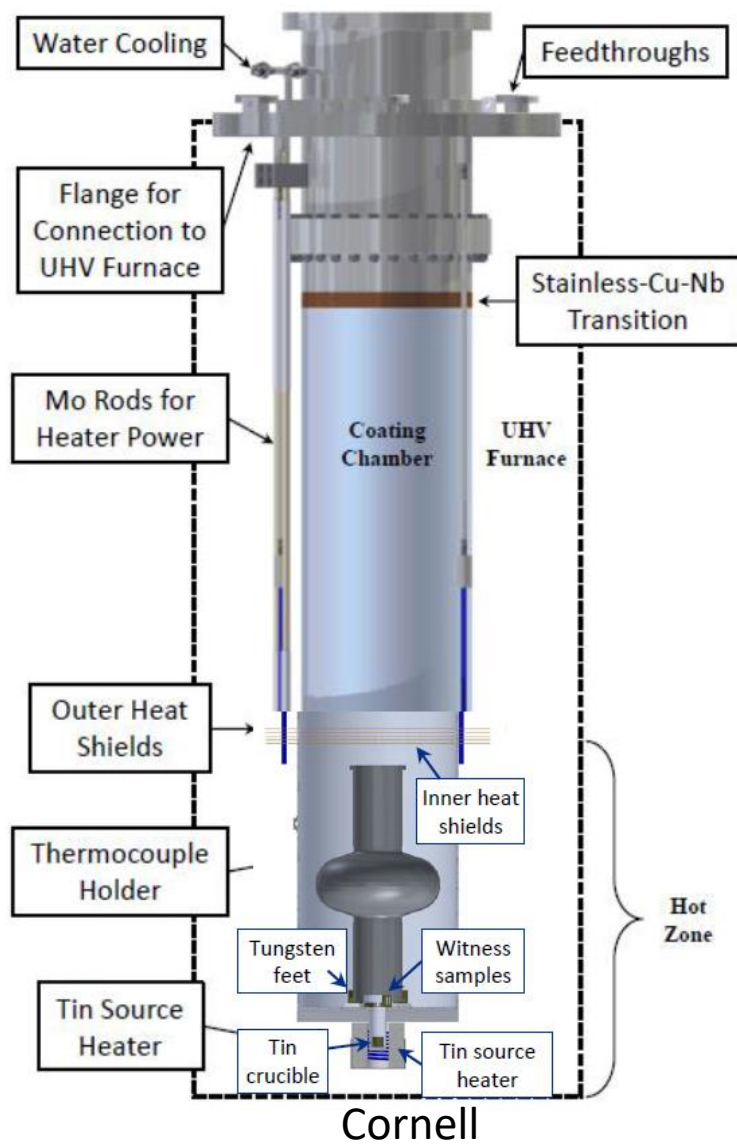


FIG. 1. All-niobium inner reaction room, consisting of the niobium cavity (1) and the niobium bottom plate (2) containing the tin source (3). The lower part of the quartz ampulla (4) is inserted into a resistance furnace.

B. Hillenbrand and H. Martens,
J. Appl. Phys. 47, 4151 (1976)

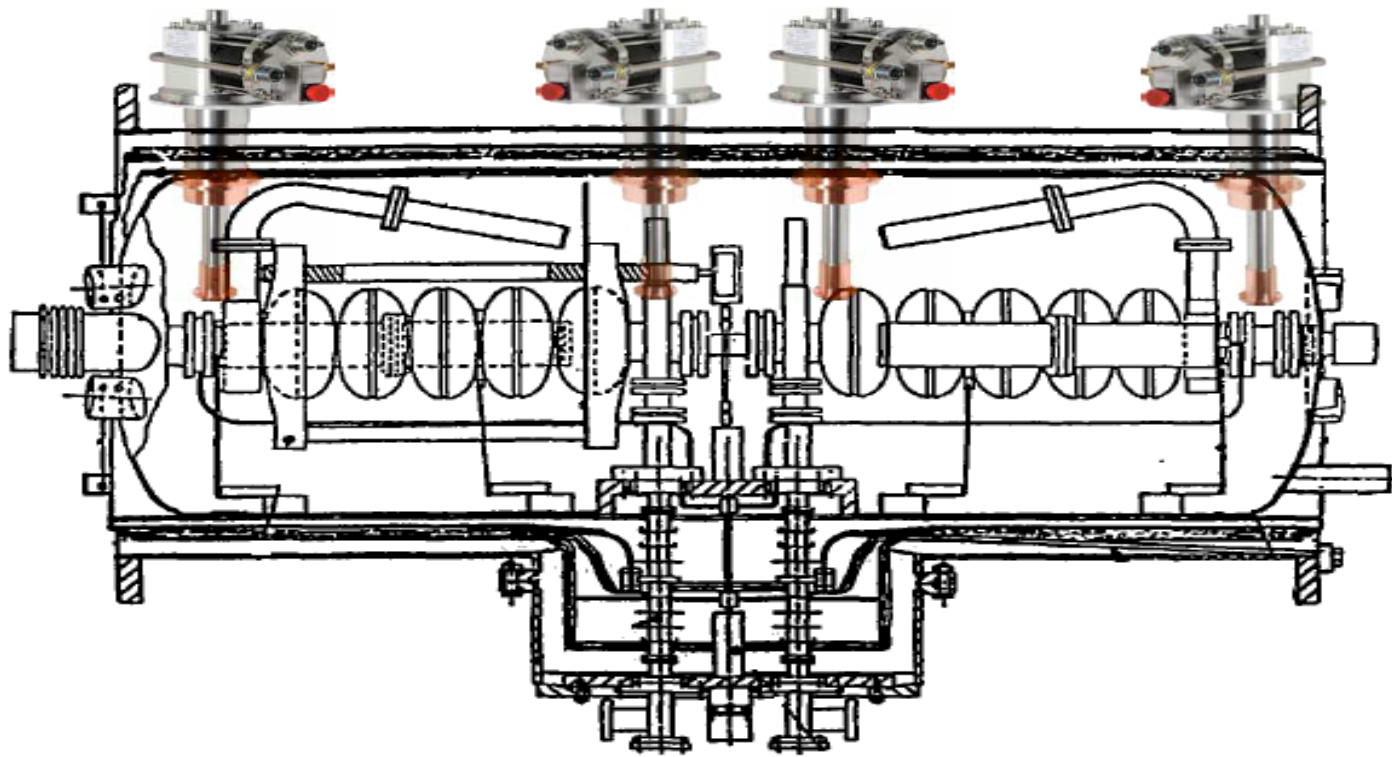


Nb₃Sn: Cornell, Jlab, and Fermilab



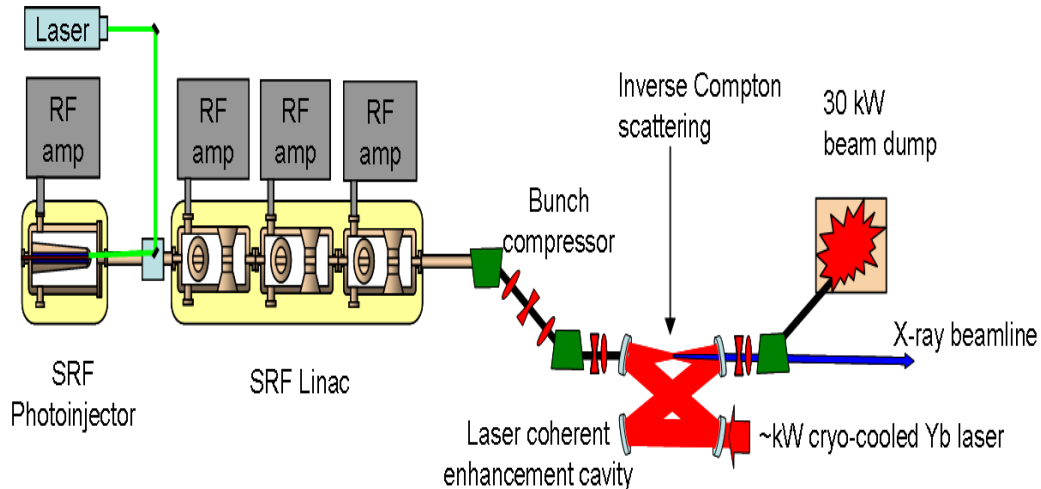
Nb_3Sn cavities cooled by cryocoolers

Cryocooler-cooled cryomodules?!

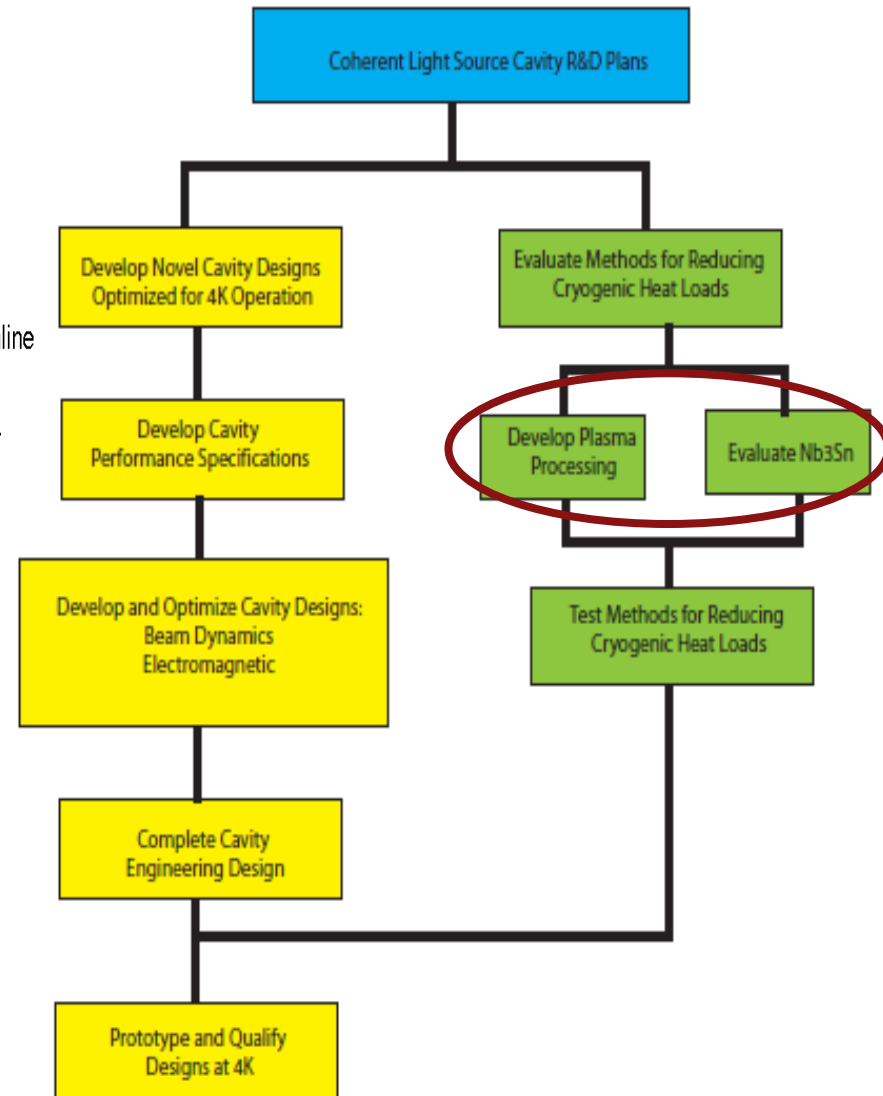


<http://www.shicryogenics.com/wp-content/uploads/2012/11/Cryocooler-Product-Catalogue.pdf>

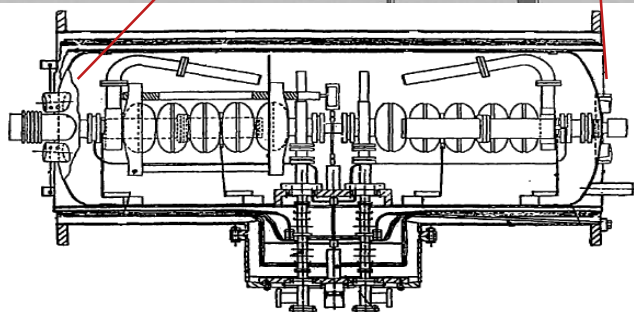
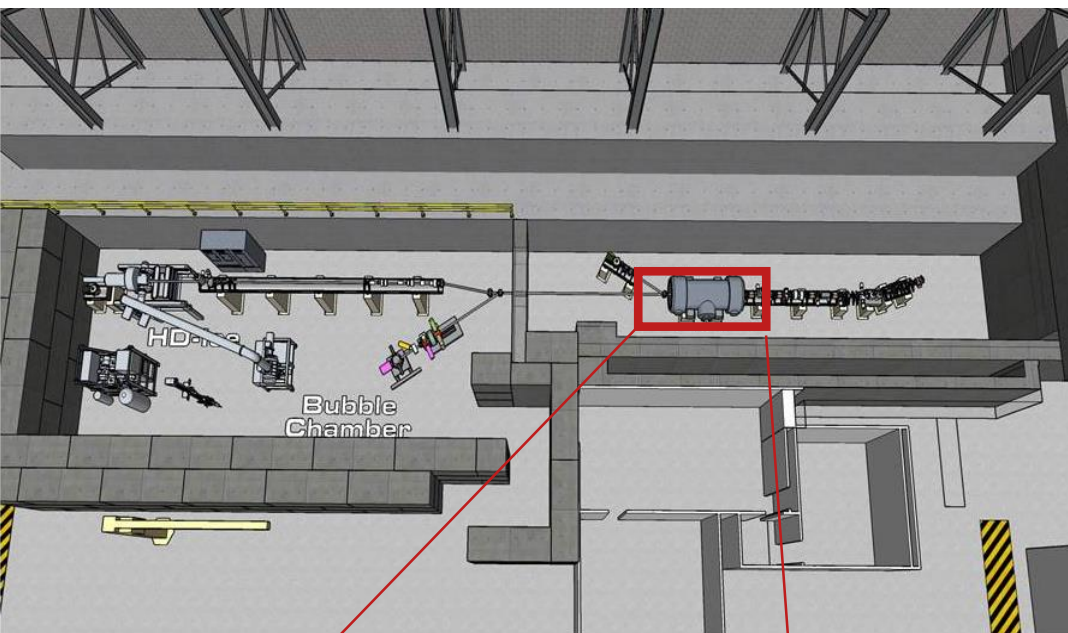
Nb₃Sn cavities for compact light sources



Injector Gain	4	MeV
Accelerator Cryomodule Gain	20-25	MeV
Number of Cryomodules	1	
Number of Cavities	2-4	
Frequency	352	MHz
Operating Temperature	4.5	K
RF Amplitude Stability	0.1?	%
RF Phase Stability	50?	fs
Cavity Type	Spoke or Elliptical	



Nb₃Sn cavities for Upgraded Injector Test Facility (UITF) @ Jlab



D. Abbott et al. , Phys. Rev. Lett. 116, 214801
 B. DiGiovine et al., Proc. AIP Conf. 1563, 239 (2013)
http://wiki.jlab.org/ciswiki/index.php/Main_Page

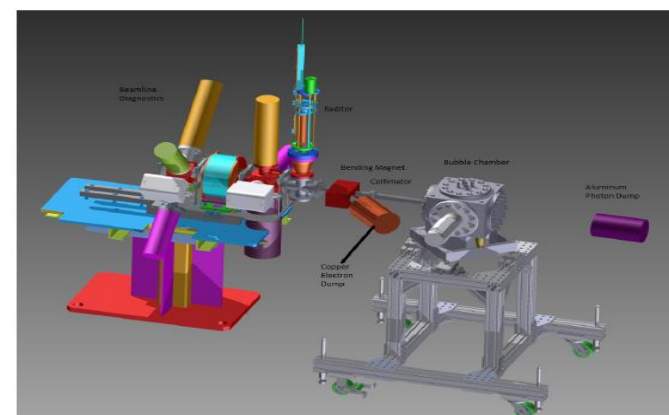
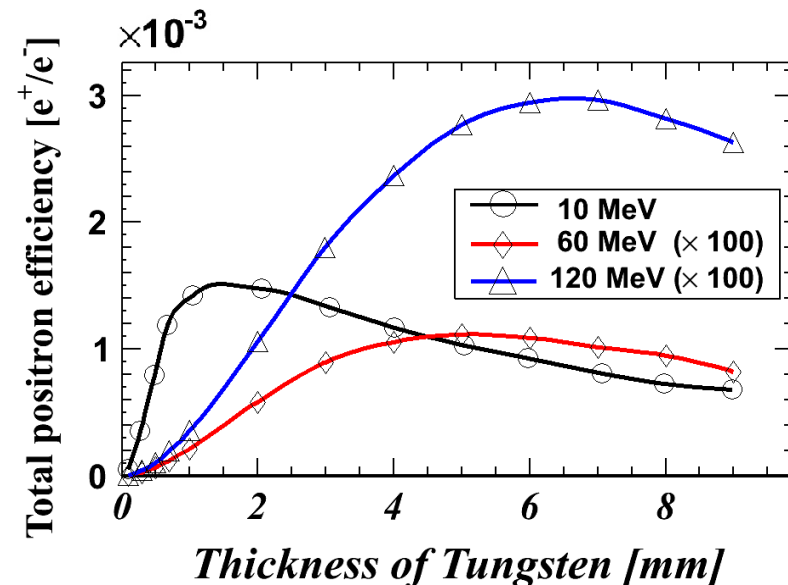
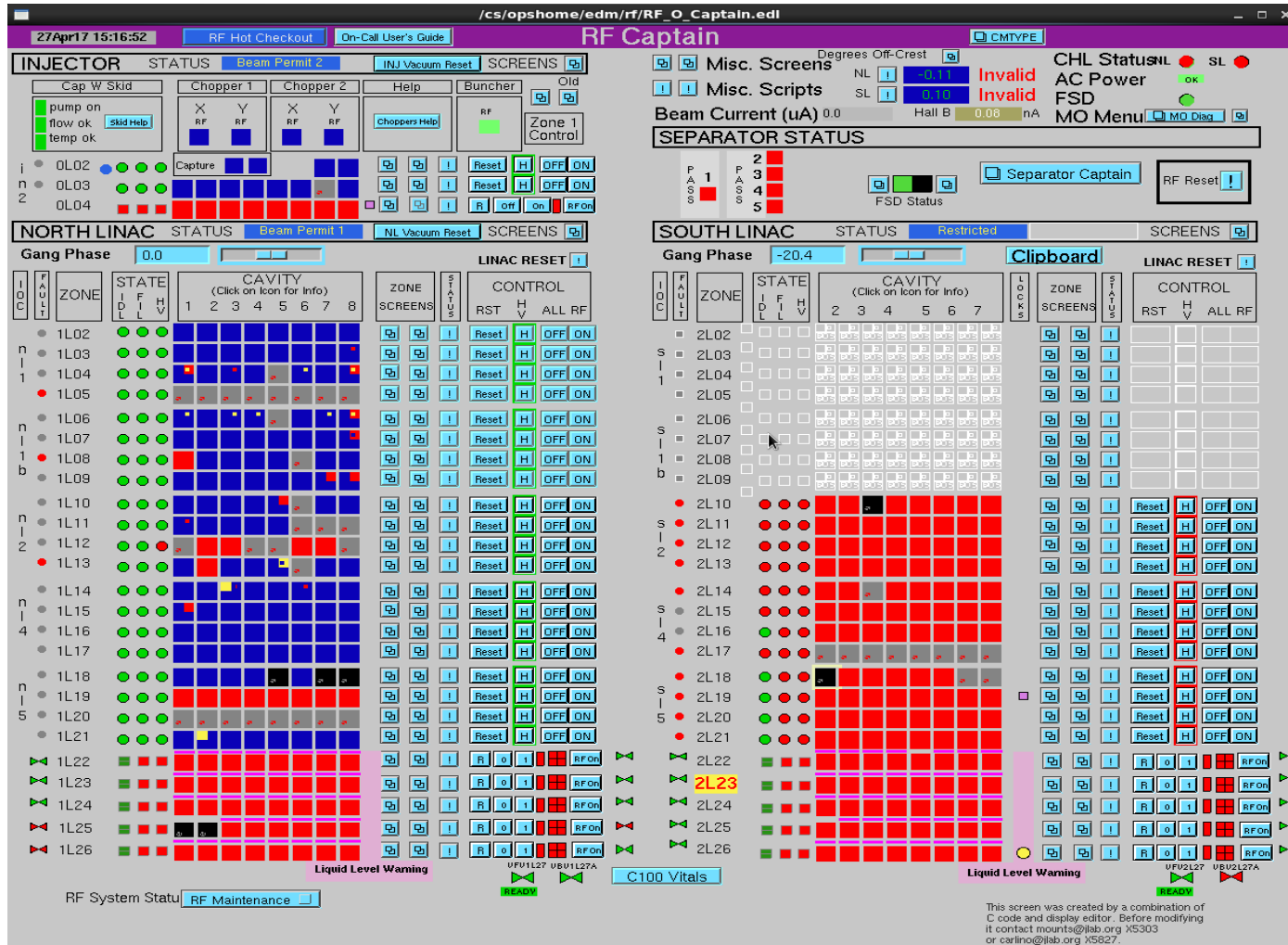


Fig. 12. Schematic of the proposed experiment.
 The photon yield that hits the bubble chamber is shown in figure 14. Here the electron beam has a kinetic energy of 8.5 MeV and is irradiating the 0.02 mm Cu radiator. Since the $^{16}\text{O}(\gamma, \alpha)^{12}\text{C}$ cross section is very steep, only photons next to the end point will produce events from this reaction.

4K vs 2K beam quality

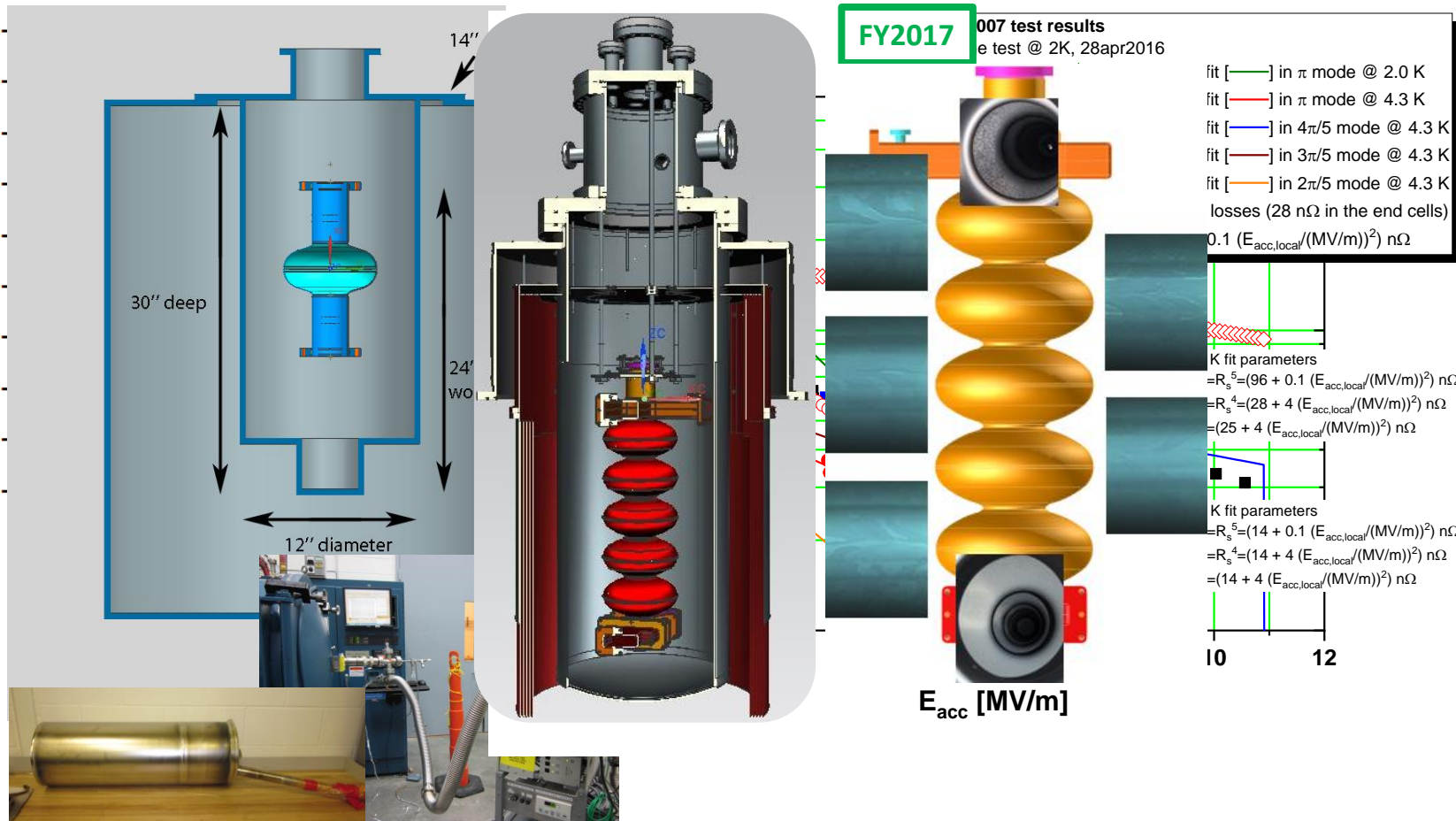
Parameter	Unit	March 23, 2016	June 17, 2016
CHL Condition	K		
Cavities	#	OL02-7,8	OL02-7,8
Gradient	MV/m	5.00, 5.32	5.00, 5.32
PSET (Crest)	deg	164.8, 83.2	-168.4, 123.6
Momentum	MeV/c	6.34	6.47
Laser Used	Hall	A	A
Max Intensity (IBC0L02)	μ A	80	60
Horizontal Normalized Emittance (MQJ0L02)	mm-mrad	<u>0.38 ± 0.01</u>	<u>0.44 ± 0.01</u>
Horizontal Beta (MQJ0L02)	m	5.21 ± 0.08	9.55 ± 0.12
Horizontal Alpha (MQJ0L02)	rad	-1.01 ± 0.01	-3.03 ± 0.04
Vertical Normalized Emittance (MQJ0L02)	mm-mrad	<u>0.34 ± 0.01</u>	<u>0.54 ± 0.01</u>
Vertical Beta (MQJ0L02)	m	2.53 ± 0.06	15.8 ± 0.1
Vertical Alpha (MQJ0L02)	rad	-0.42 ± 0.01	-4.39 ± 0.02
Horizontal Profile Scan (IHA2D00)	mm	2.35 ± 0.02	1.46 ± 0.02
Momentum Spread (dp/p)	%	0.22%	0.14%
Energy Spread (dE/E)	keV	<u>14</u>	<u>9</u>

4K CEBAF test



N. Hasan, C. Mounts, W. Oren, A. Solopova, M. Wright, M. Drury, J. Grades, R. Kazimi, M. Poelker, T. Powers, J. Preble, R. Suleiman, Y. Wang, M. Wright, A. Hutton, H. Areti et al.

Jlab Nb₃Sn development timeline



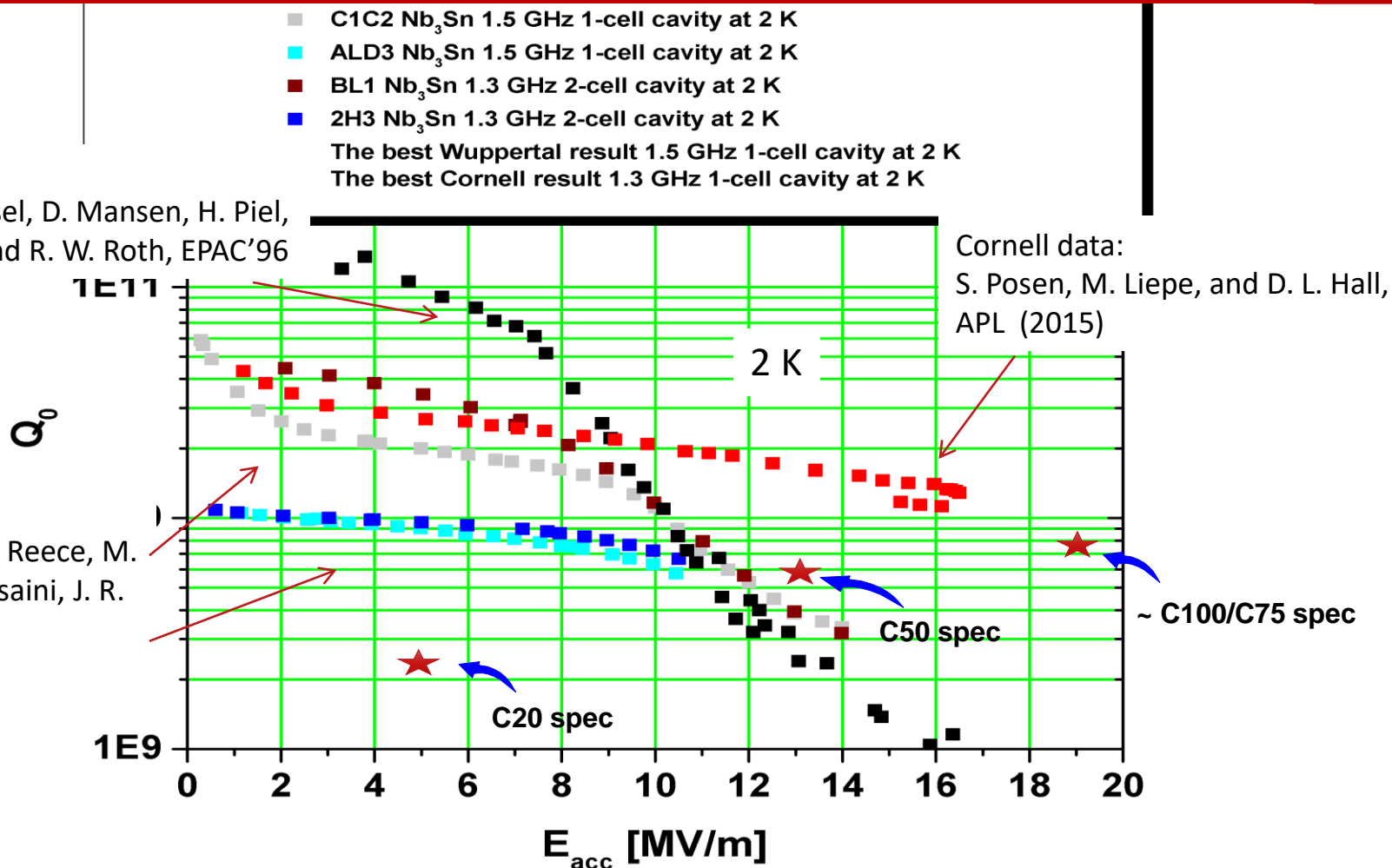
Present single-cell work

Wuppertal data:

G. Müller, P. Kneisel, D. Mansen, H. Piel,
J. Pouryamout, and R. W. Roth, EPAC'96

JLab data:

G. Ereemeev, C. E. Reece, M.
J. Kelley, U. Pudasaini, J. R.
Tuggle, SRF'15

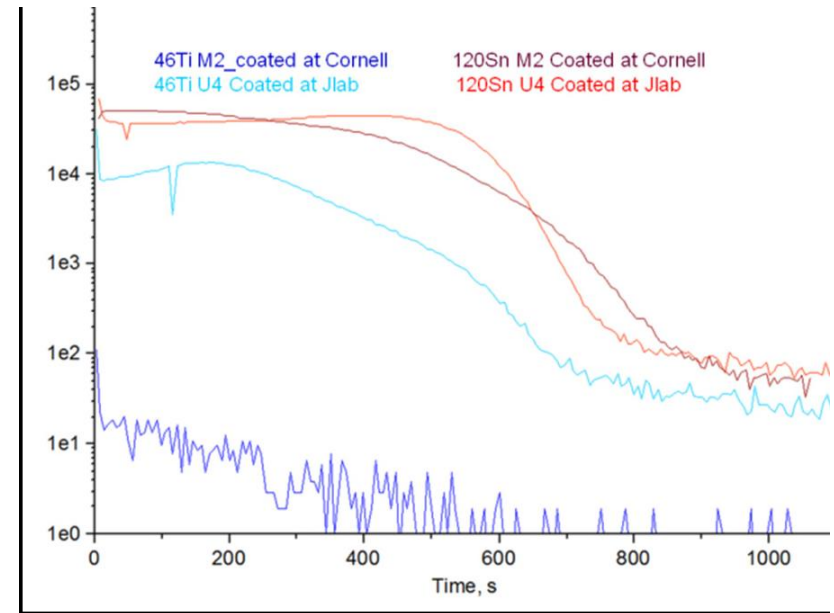
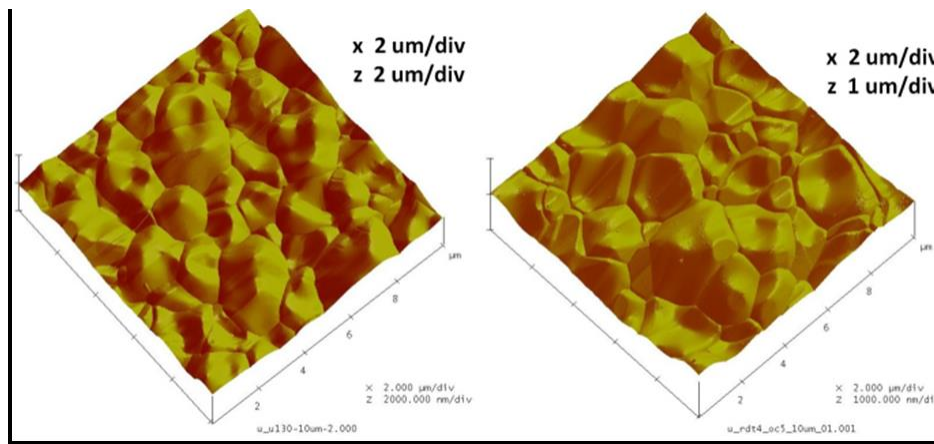


05.03.16

**DOE's Office of Science Selects 49 Scientists to Receive
Early Career Research Program Funding**

Ereemeev, Grigory V., Thomas Jefferson National Accelerator Facility, Newport News, VA, "Formation of Superconducting Nb₃Sn Phase for Superconducting Radio Frequency (SRF) Cavities," selected by the Office of Nuclear Physics.

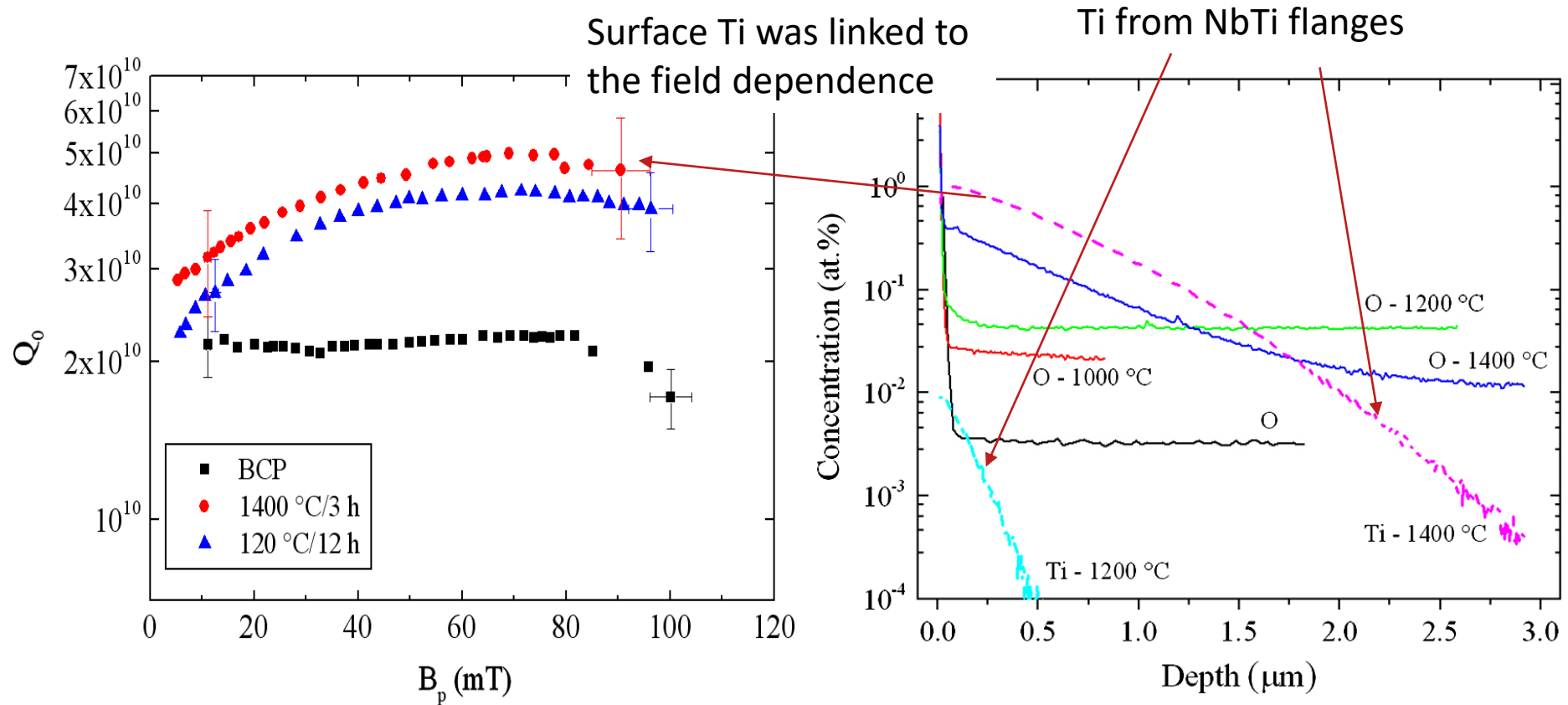
Titanium hypothesis



- 20 -

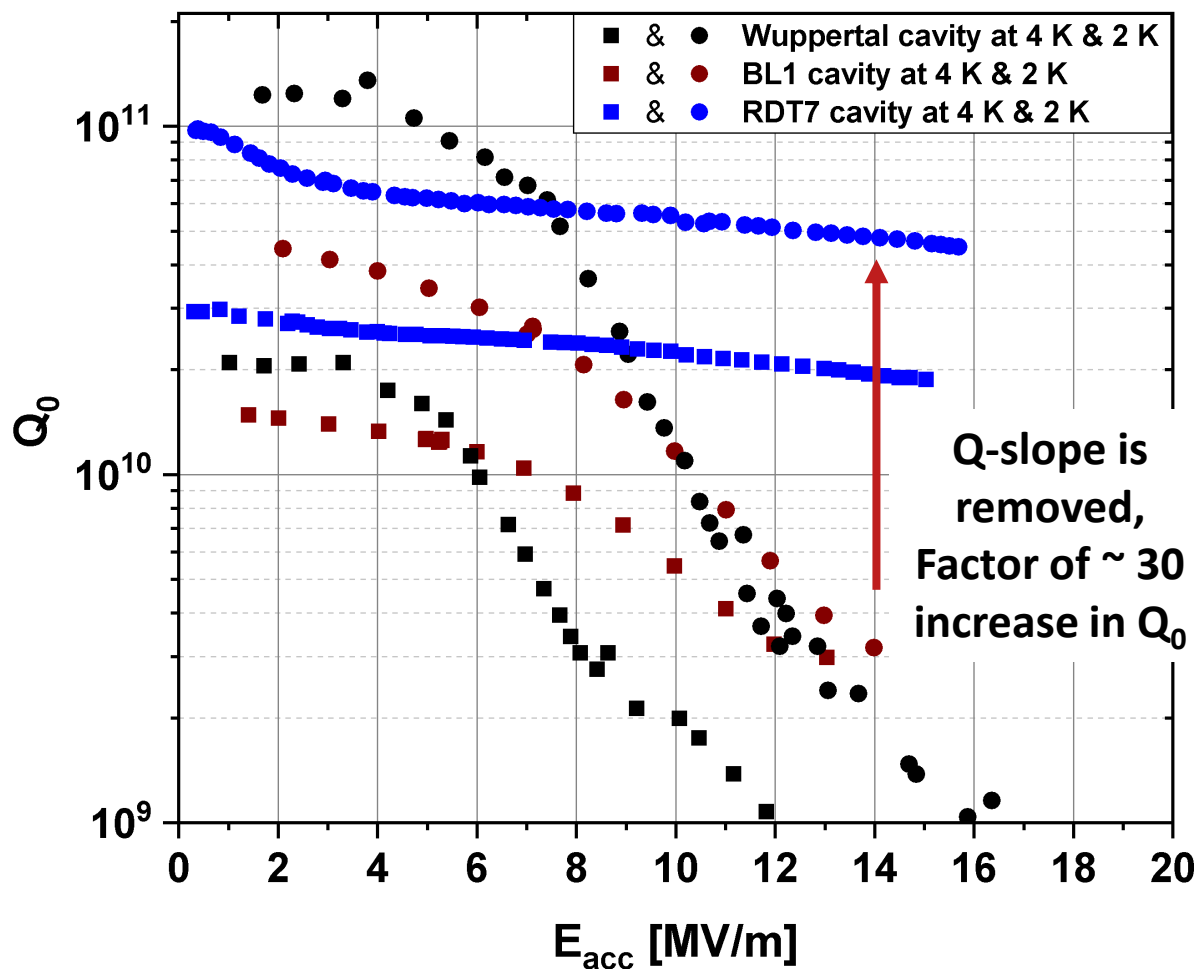
Zum anderen wurde zur Reduktion des Sauerstoffpartialdrucks im Ofeninneren der Resonator außen mit einer 0.5 mm dicken Titanfolie ummantelt. Dies führte während der Nb_3Sn -Beschichtung zu einer Titanbeschichtung der Resonatoraußenfläche. Eine geringe ^{keine} Verunreinigung ^{kontamination} der innen aufwachsenden Nb_3Sn -Schicht durch hineindiffundieren- des Titan wird man praktisch kaum vermeiden können (siehe Kap. II.3). Dieser Effekt wird aber als unkritisch angesehen, da nach Ref. 71 Titananteile von 5 % nur zu einer T_c -Reduktion von weniger als 0.2 K führen. Zur Vermeidung von Keimbildungsproblemen

Titanium hypothesis



Effect of high temperature heat treatments on the quality factor of a large-grain superconducting radio-frequency niobium cavity, P. Dhakal et al., Phys. Rev. ST Accel. Beams 16, 042001, 2013

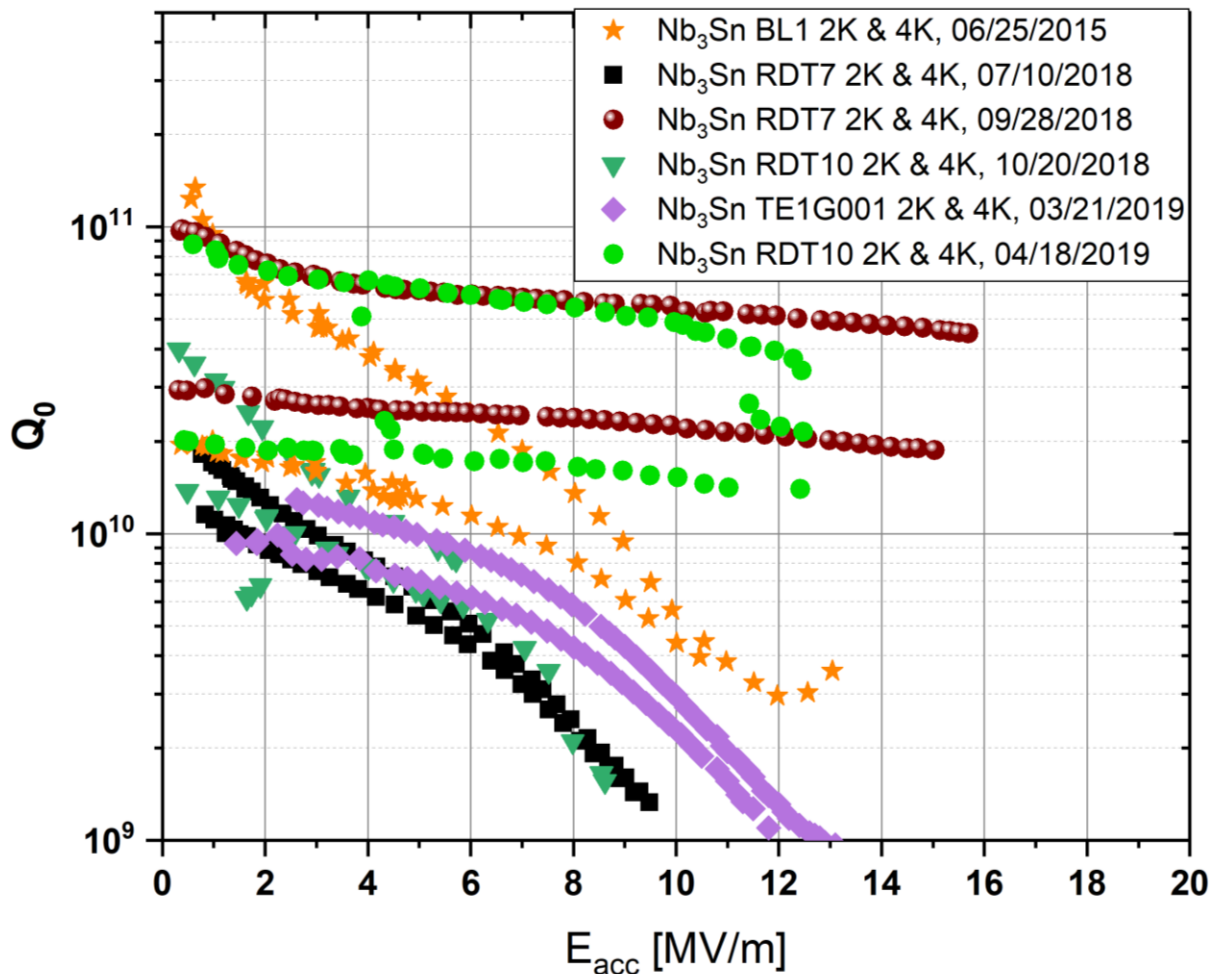
Recent data after the coating system upgrade



- Following system upgrade, Q-slope free Nb₃Sn-coated cavity were observed
- Q_0 improved at all fields
- At low fields, Q_0 reached 10^{11}
- $Q_0 \sim 5 \cdot 10^{10}$ @ $E_{acc} = 15$ MV/m
- Cavities are still coated in “Siemens” configuration, i.e., no secondary heater for the tin source
- The cavity had NbTi flanges replaced with Nb flanges

G. Ciovati, I. Parajuli, U. Pudasaini

Current data

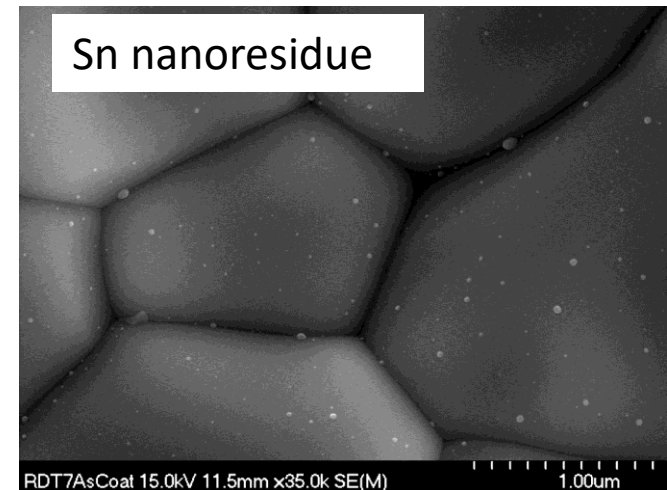
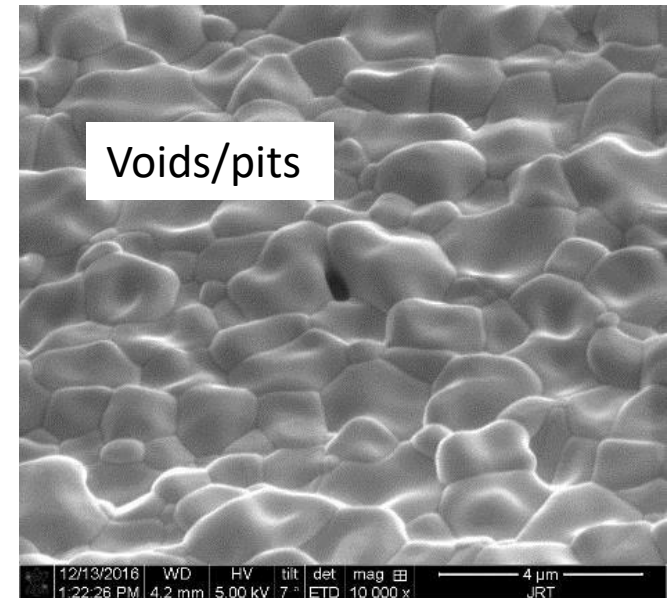
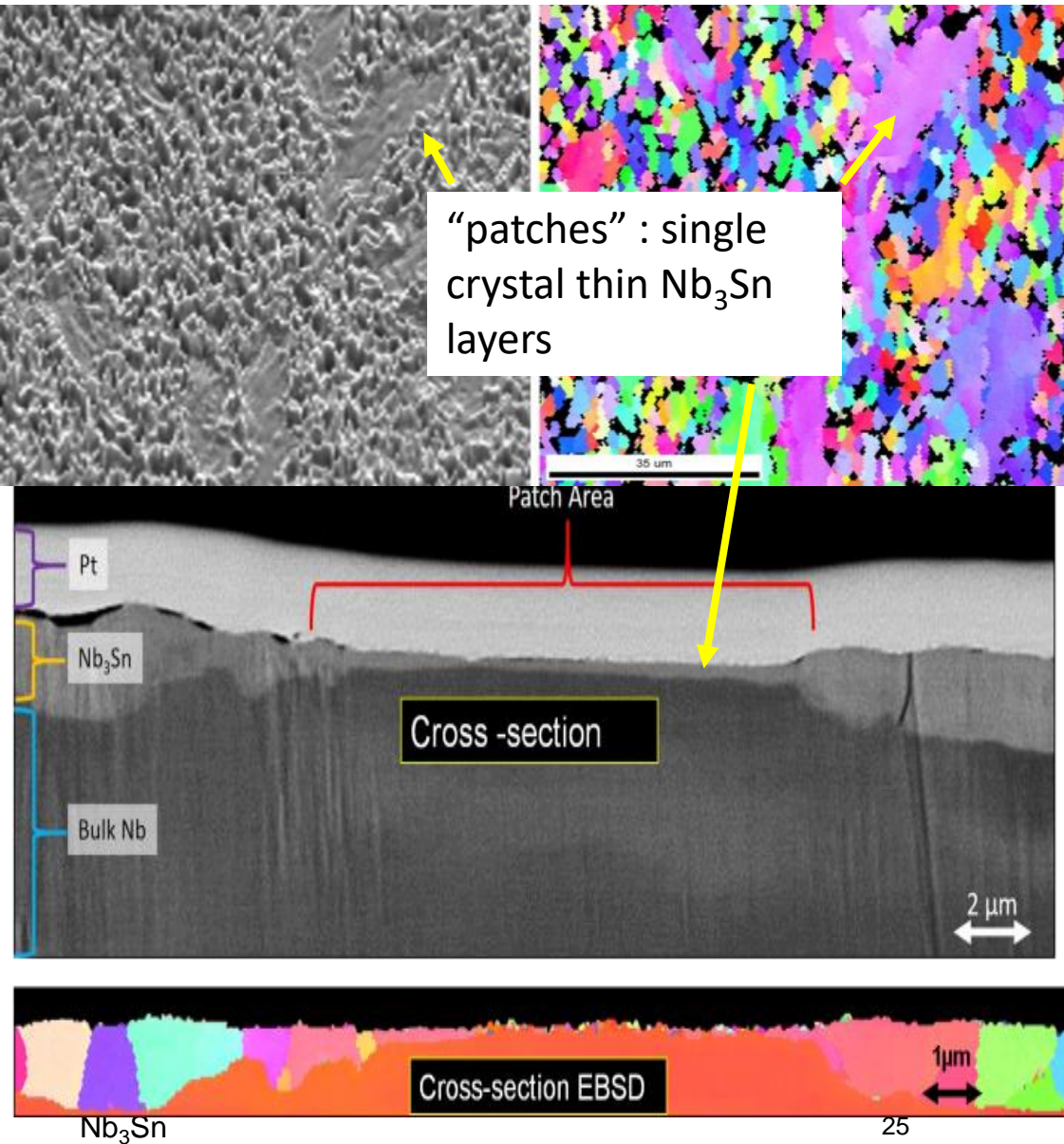


- Q-slope free Nb_3Sn -coated cavity was reproduced on another cavity
- Consistent Q_0 between Q-slope free cavities
- Q-slope limited performance for some coatings was linked to variation in Sn source; studies are ongoing
- RDT7, RDT10 & TE1G001 had NbTi flanges replaced with Nb flanges

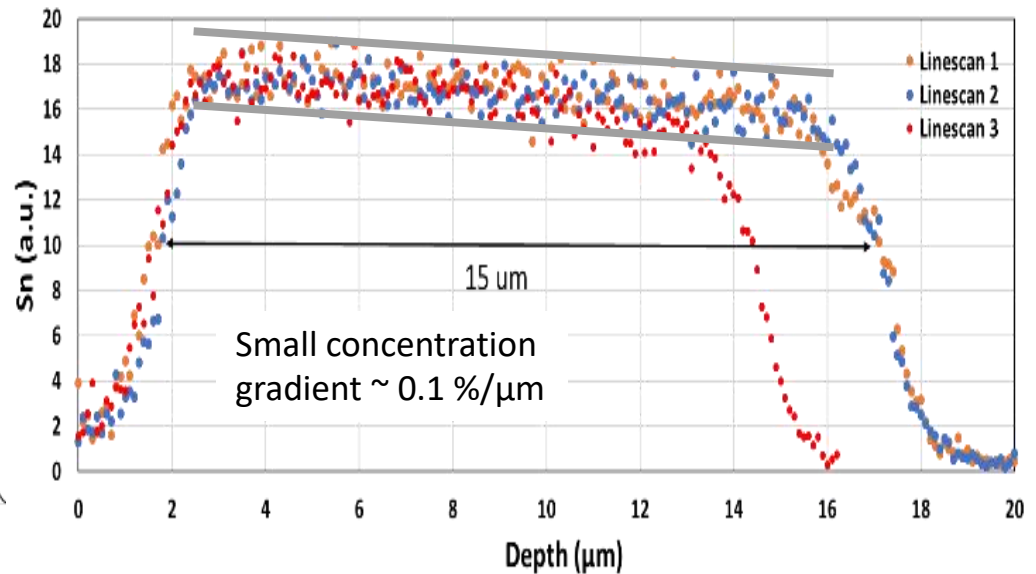
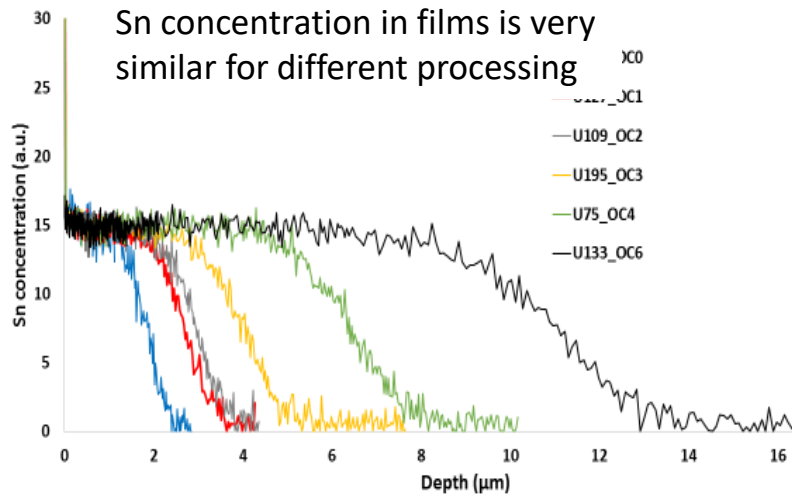
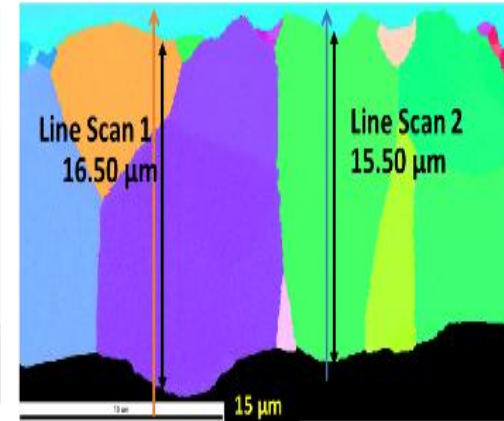
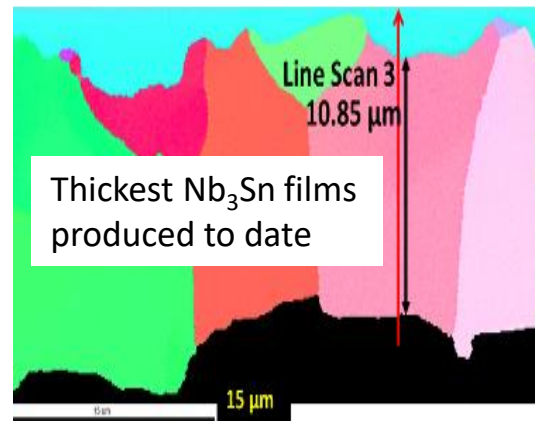
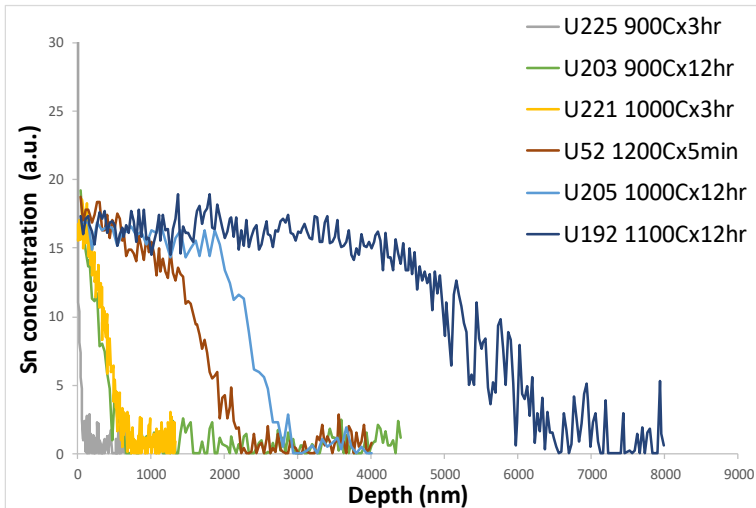
U. Pudasaini

Nb₃Sn growth and defects

U. Pudasaini, J. Tuggle

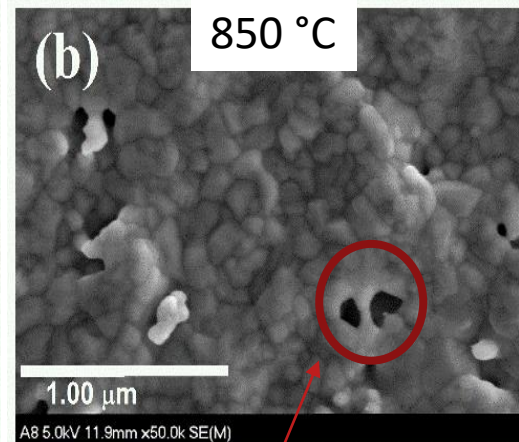
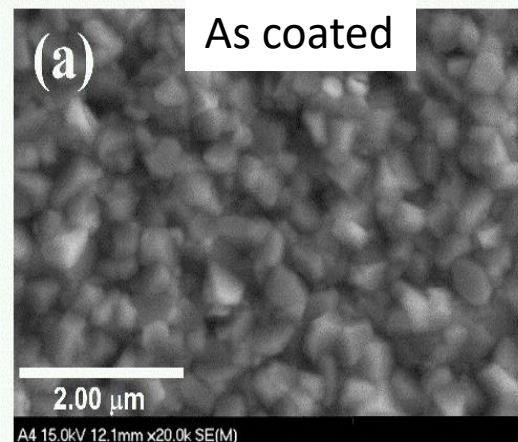
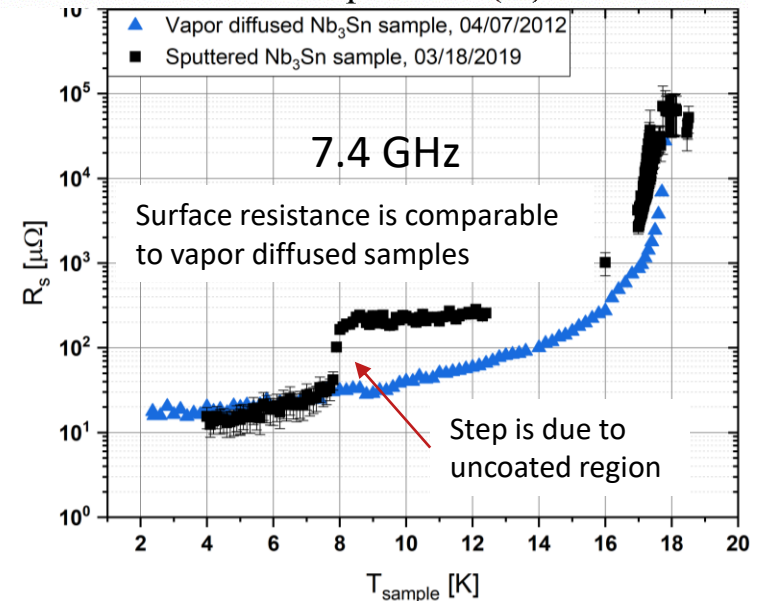
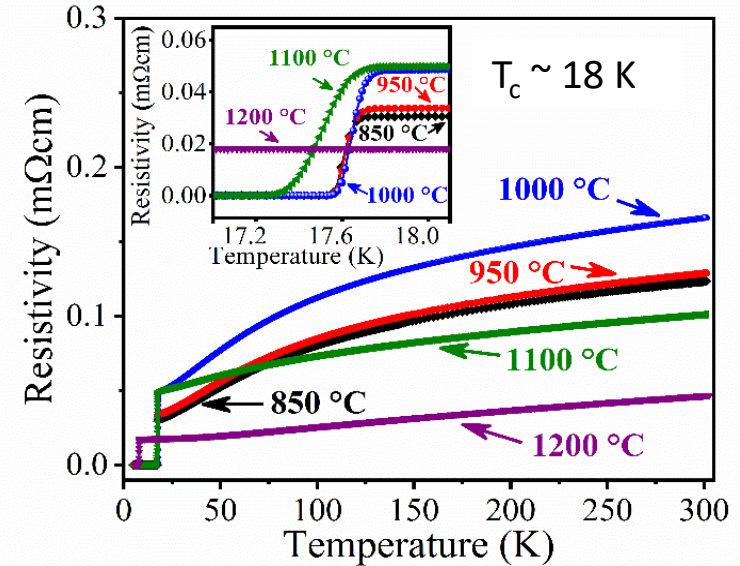
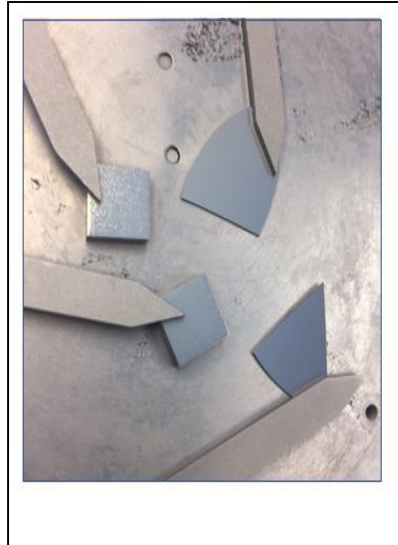
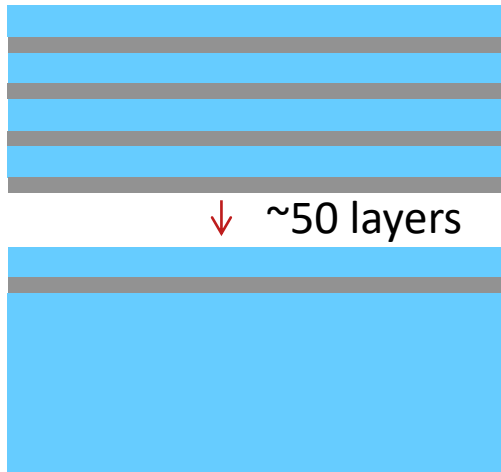


Concentration gradients



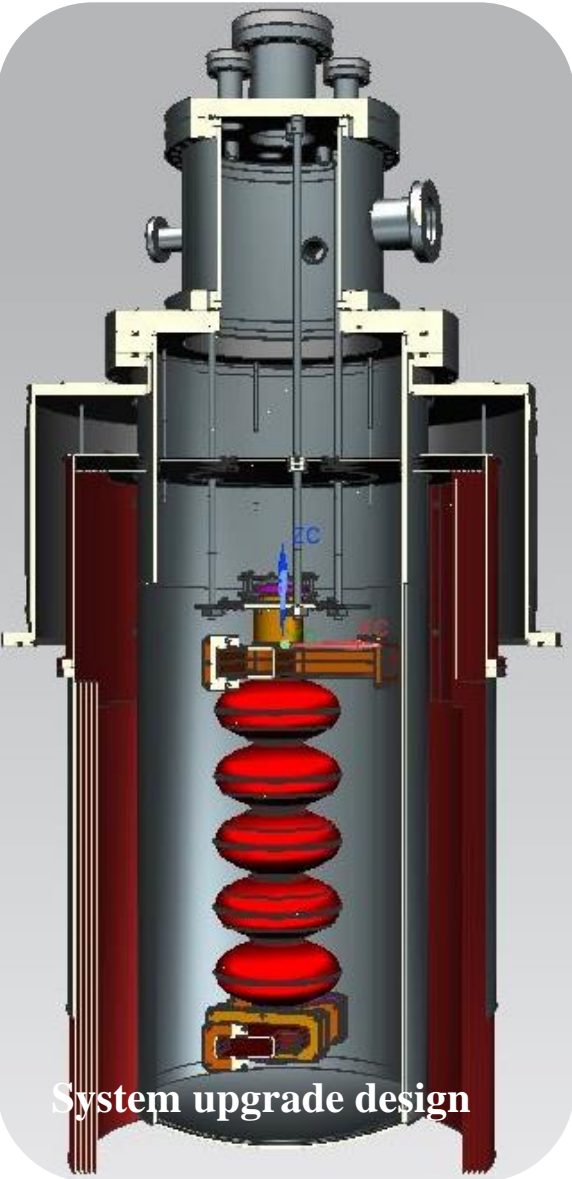
Concentration gradients, stoichiometry, and sputtering

The goal : precise control of Sn content in Nb_3Sn films

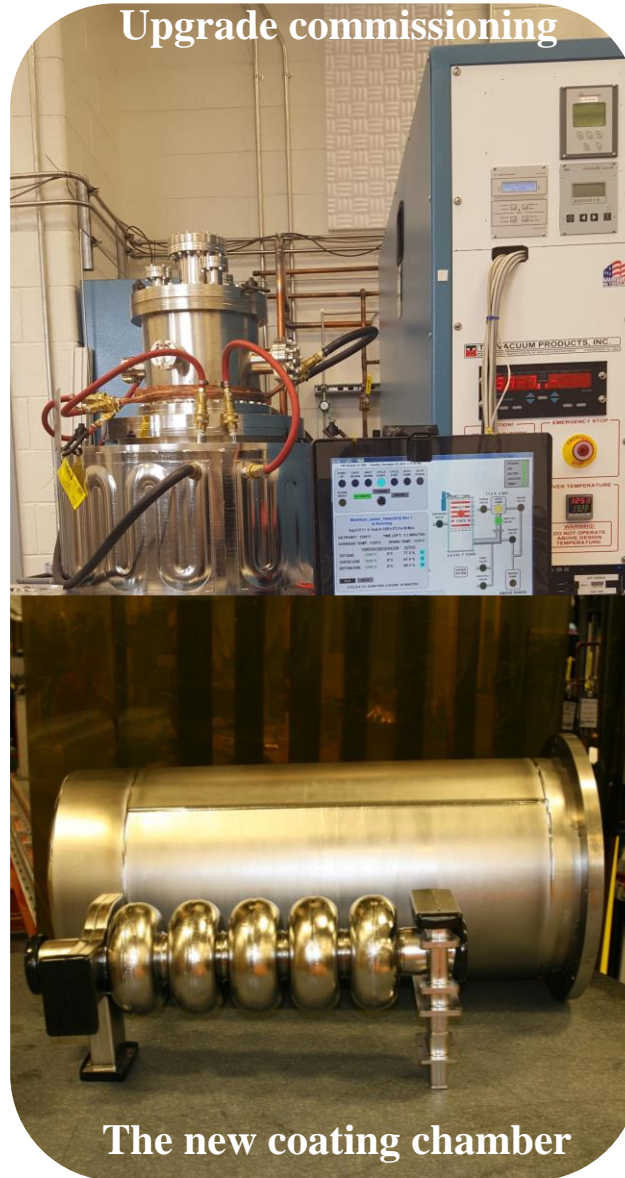


Needs to reduce/eliminate voids,
e.g., by adjust heat treatment

Application to 5-cell cavities



Nb_3Sn



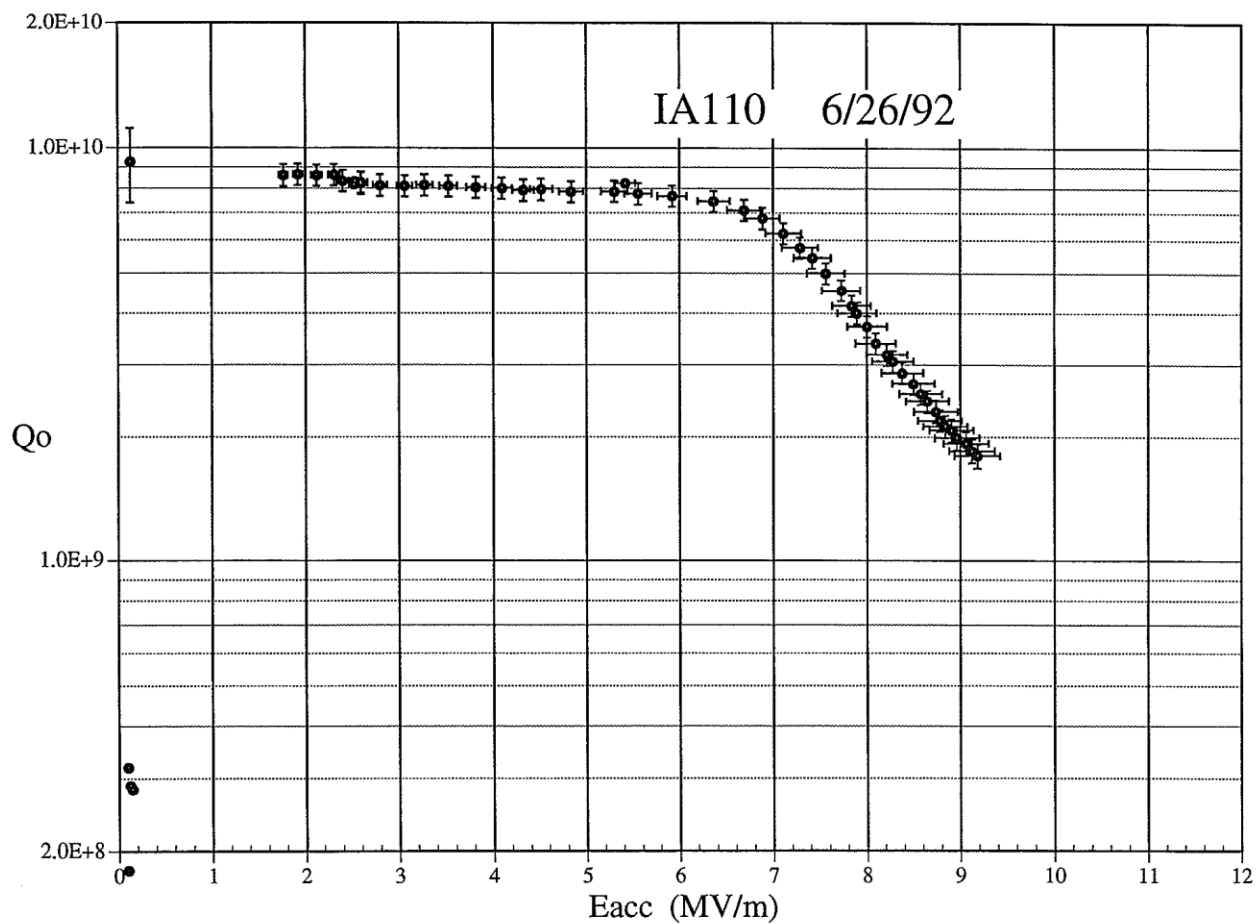
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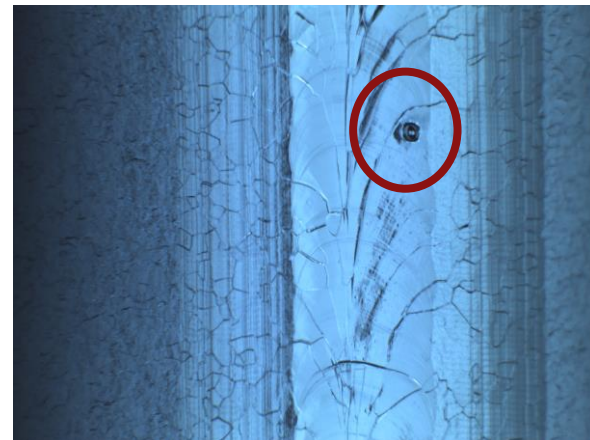


Jefferson Lab

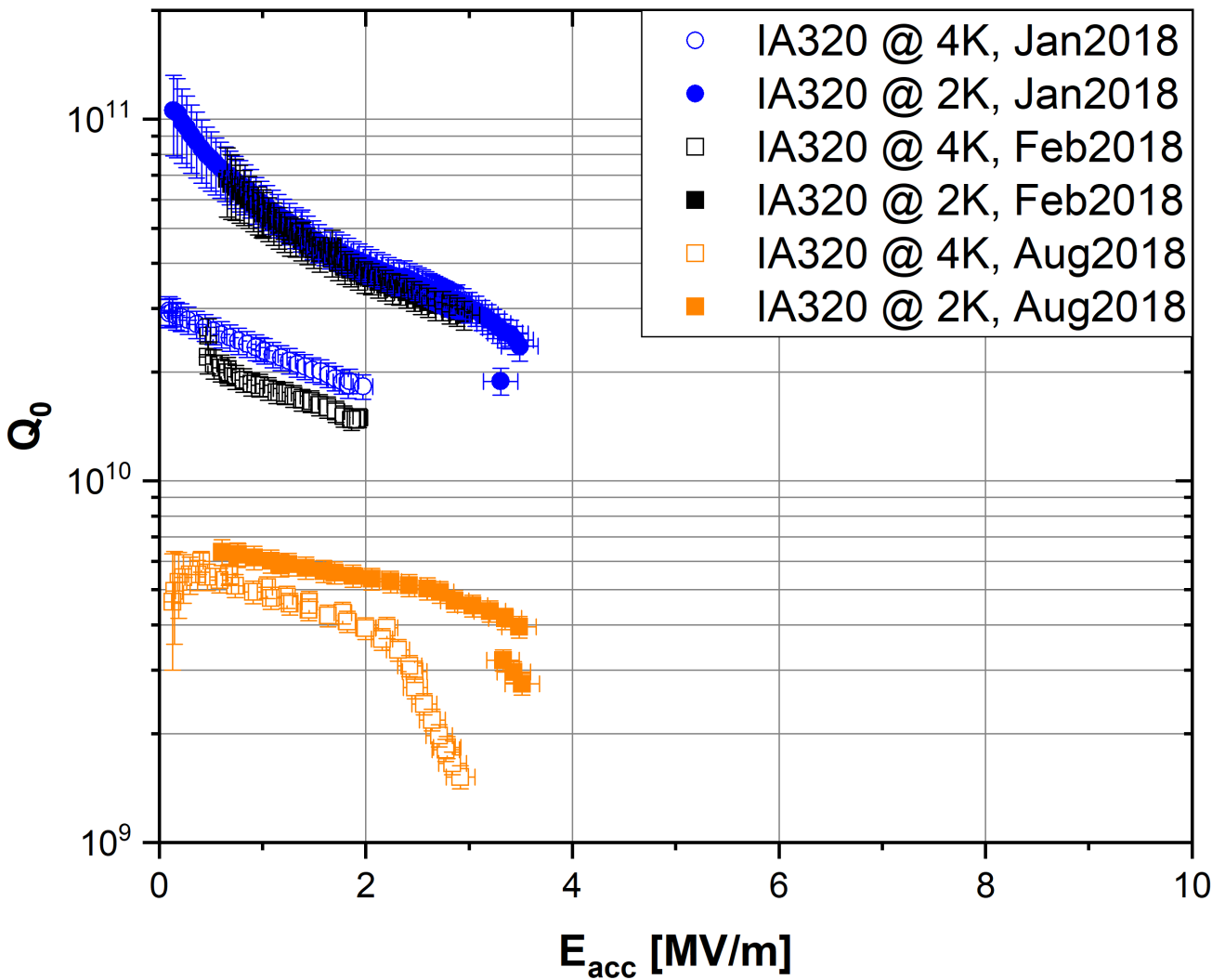
CEBAF 5-cell cavities



Tested by C. Reece!

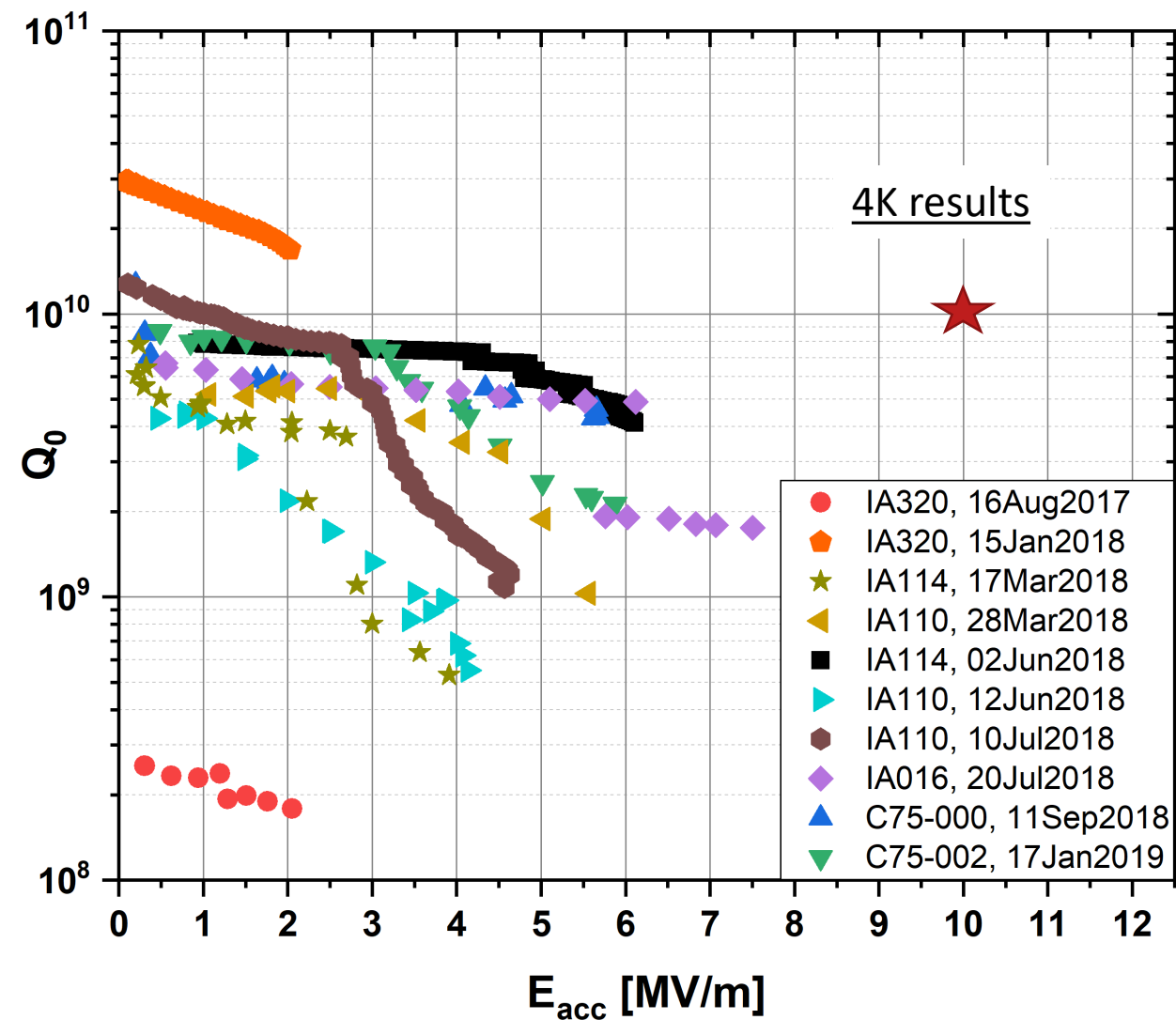


5-cell cavity coating results

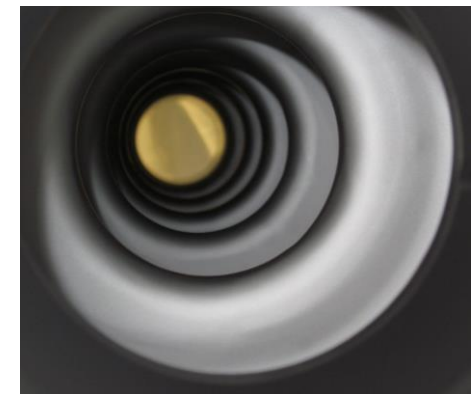


- The first CEBAF cavity coated in the upgraded system
- The cavity limited at $E_{acc} = 11$ MV/m in the baseline test before coating
- Results are shown for the coating #8 done in Nov. 17
- Coated cavity had high Q_0 ($\sim 10^{11}$), but a strong Q -slope
- Re-tests after December 2017 to see if there is any degradation
- Clear degradation in August test...why?

5-cell cavity coating results

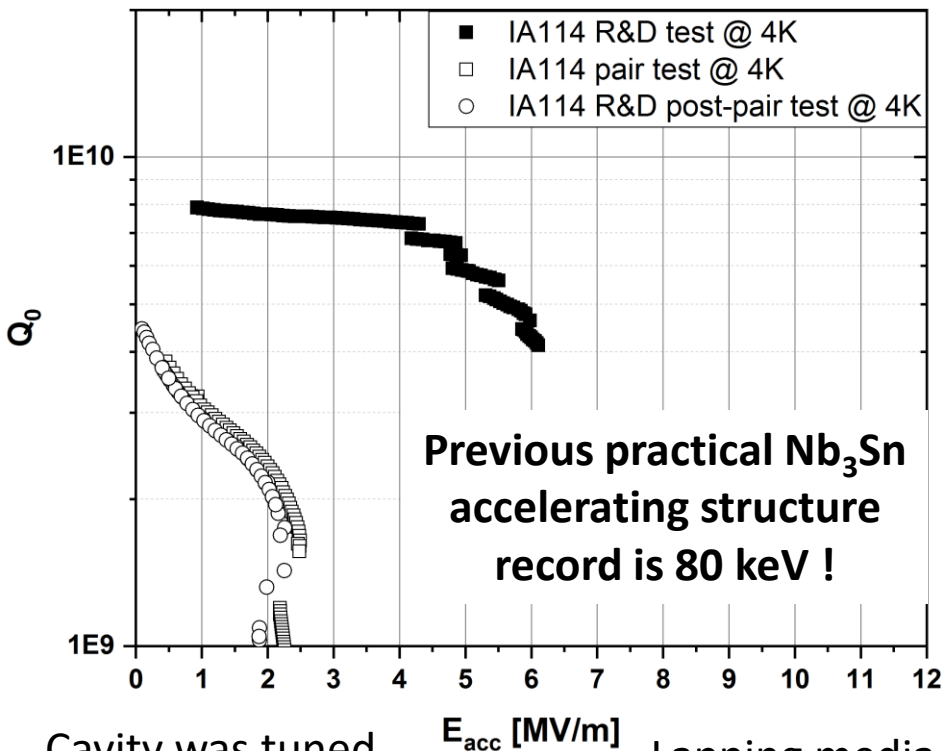


**Uniform coating, no
obvious asymmetry!**



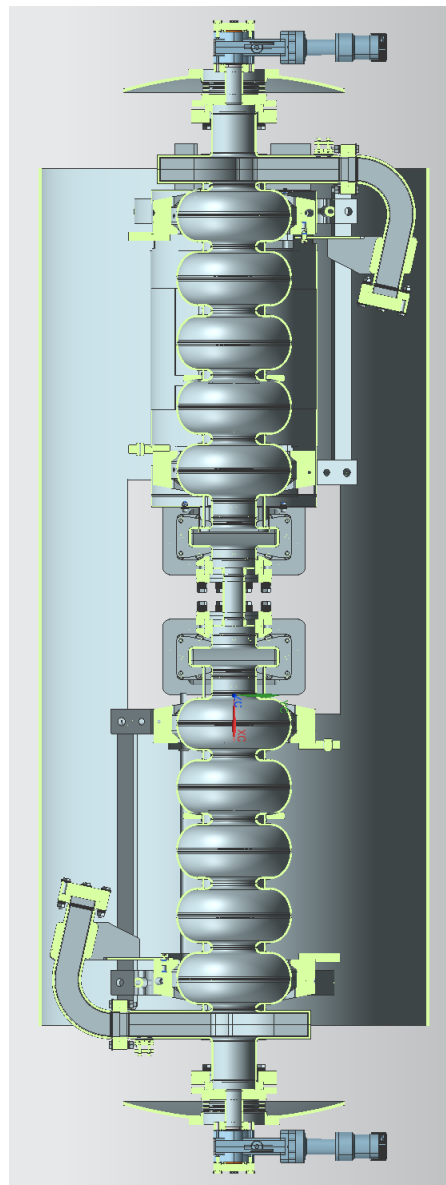
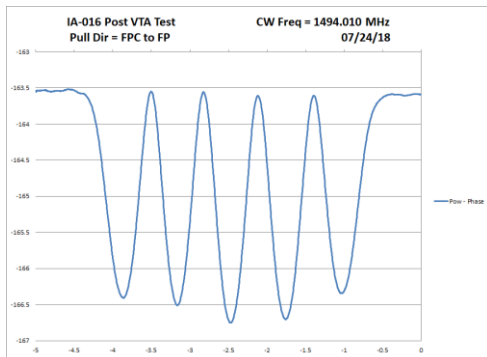
U. Pudasaini

Pair work and results

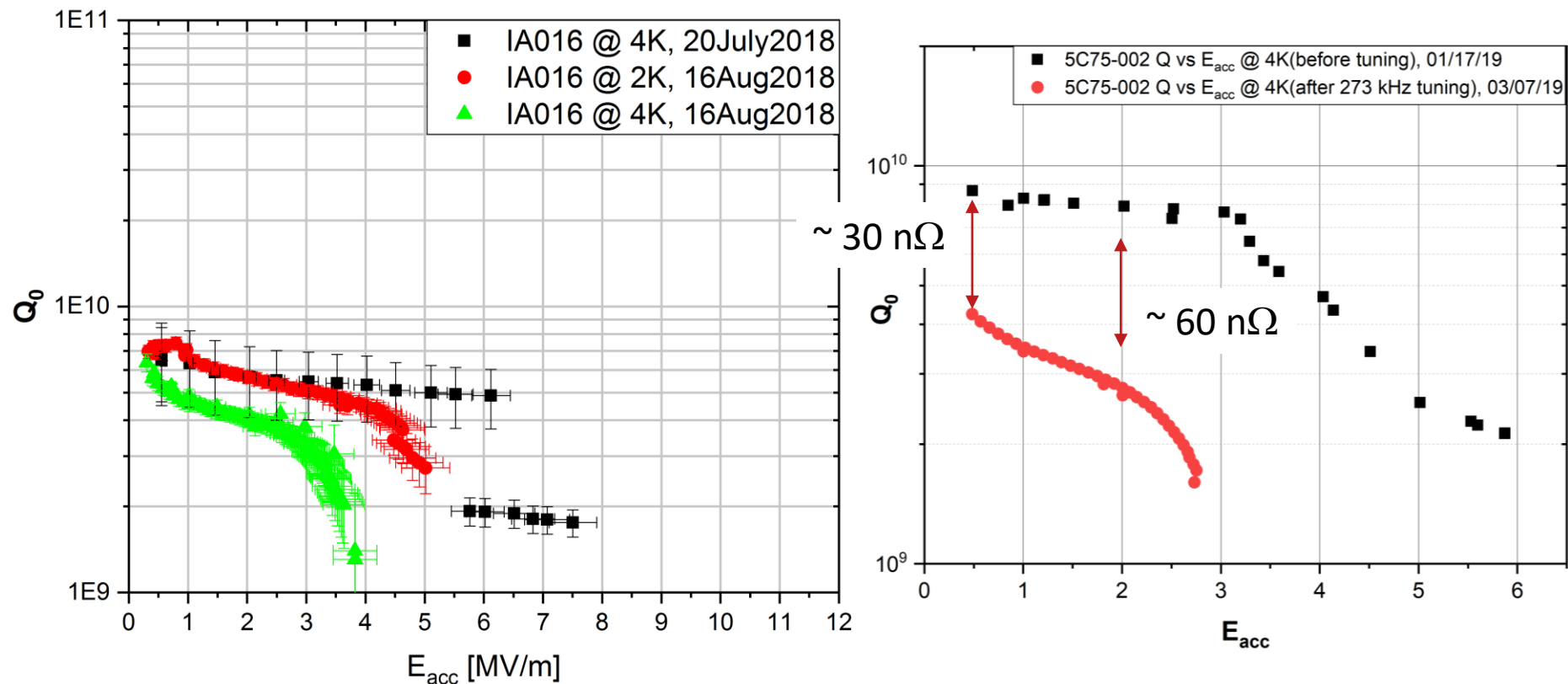


Cavity was tuned several times

Lapping media after flange polish

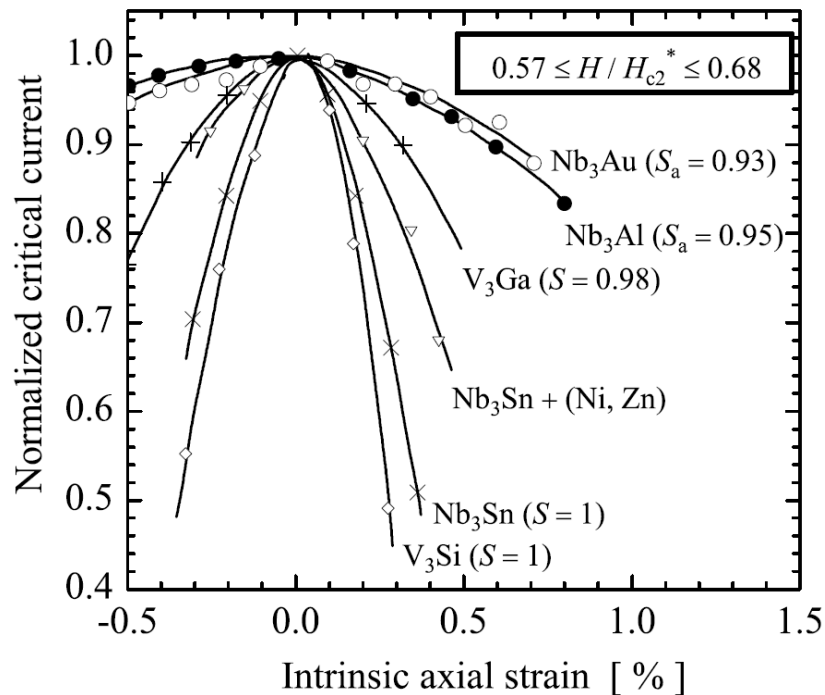


Pair work and results

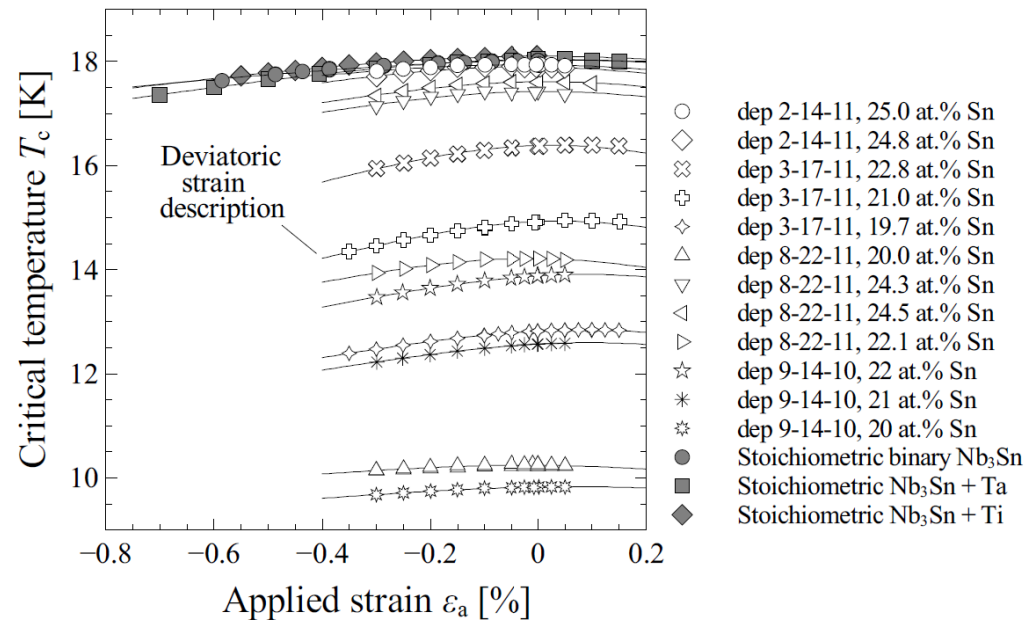


Quality factor and quench degraded after the cavity was tuned by about 200 kHz down. Tuning added field-dependent surface resistance, which increase by about 30 n Ω at low fields

Strain sensitivity



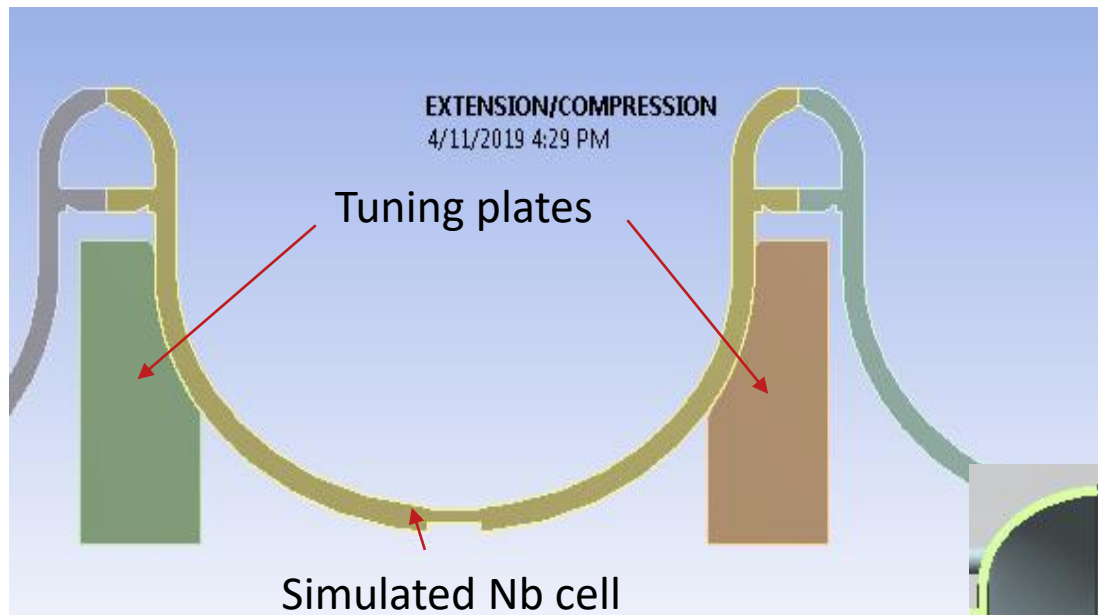
Degradation of critical current as a function of strain for some materials



Dependence of the critical temperature on strain in Nb₃Sn

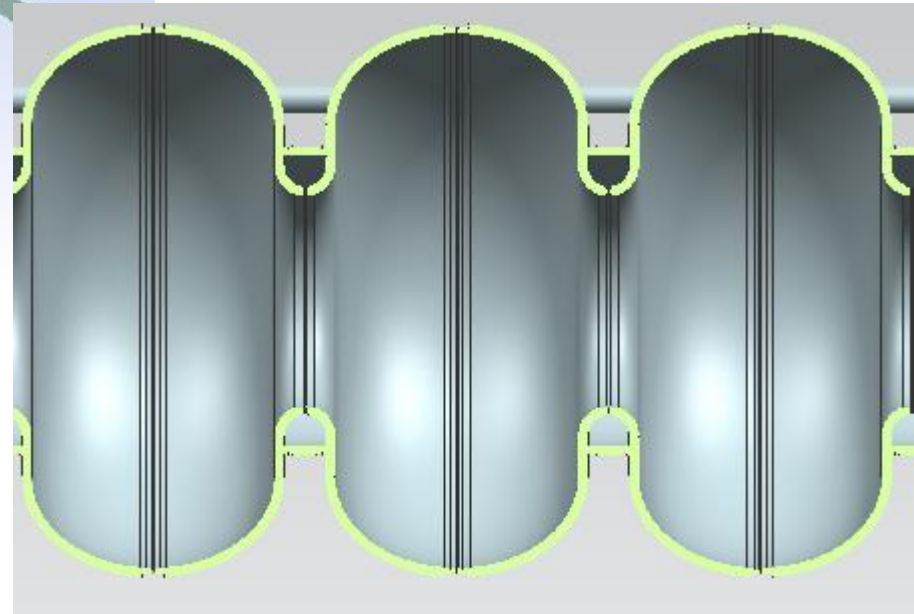
A. Godeke, Ph.D. dissertation
M. Mentink, Ph.D. dissertation

Tuning simulation



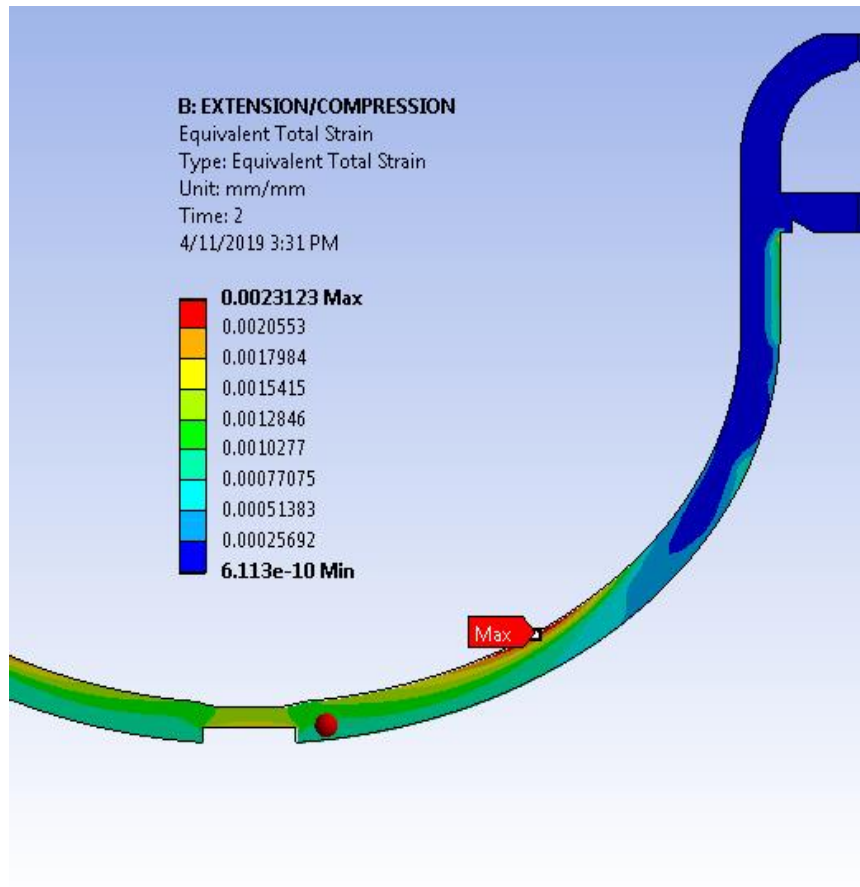
1 mm change in the cavity length corresponds to ~ 300 kHz of the frequency change

The goal was to simulate compression and extension of the center cell. The cavity needs to be squeezed/stretched beyond the desired frequency change in order to achieve the desired plastic deformation.

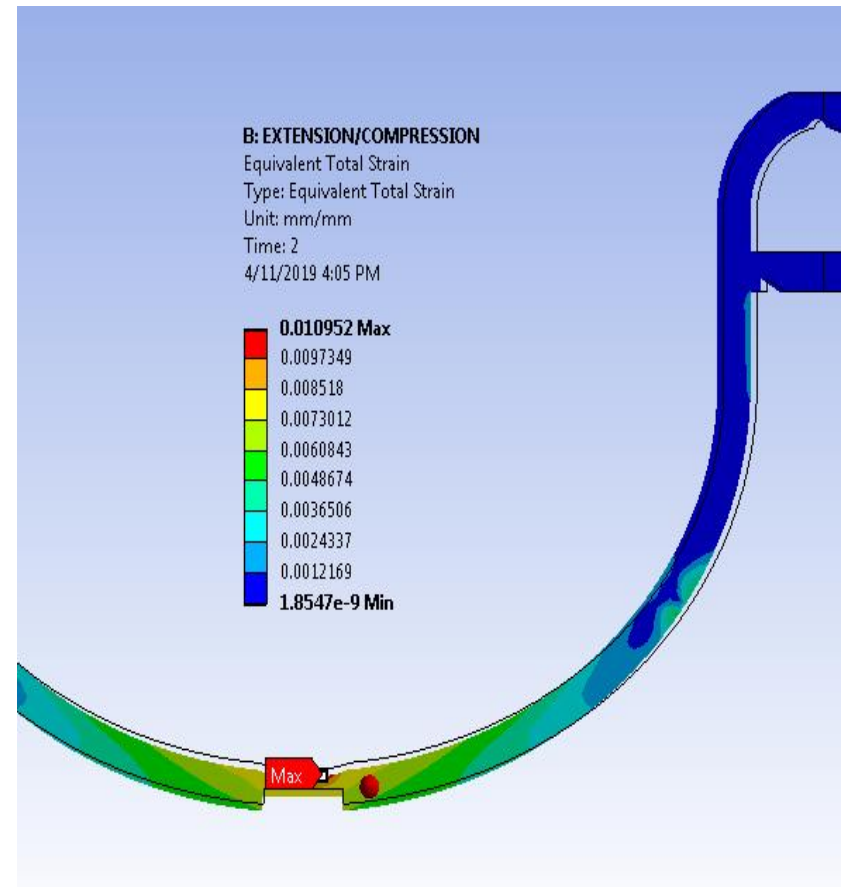


Tuning simulation

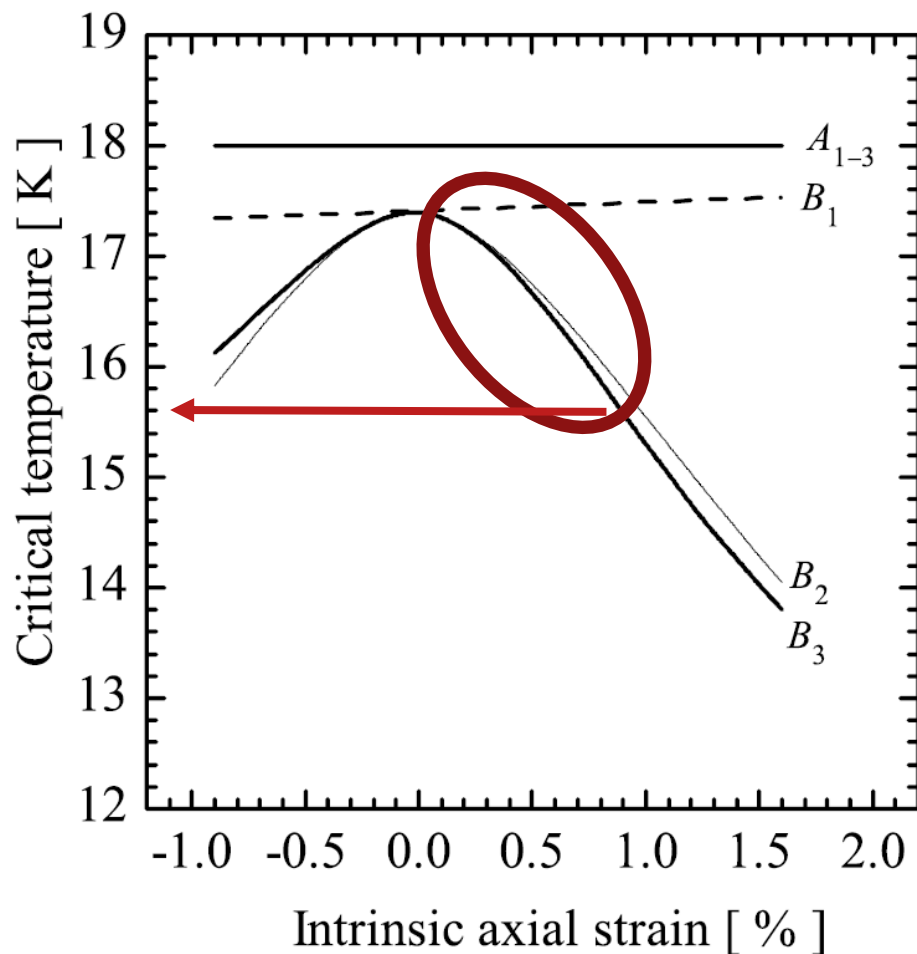
Equivalent Total Strain : 0.71 mm jaw
compression yields 0.25 mm deformation



Equivalent Total Strain: -1.445 jaw
compression yields 1 mm deformation

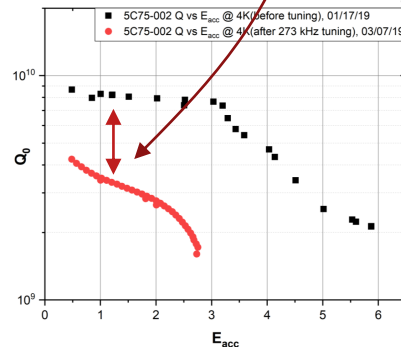
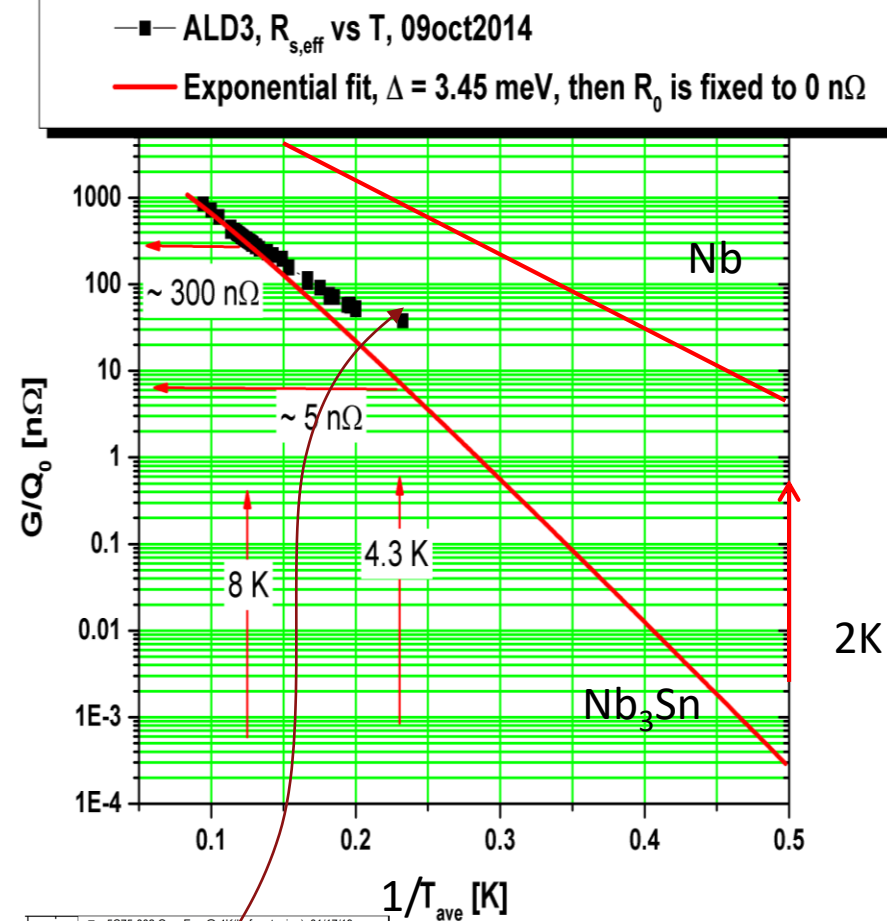


Tuning simulation



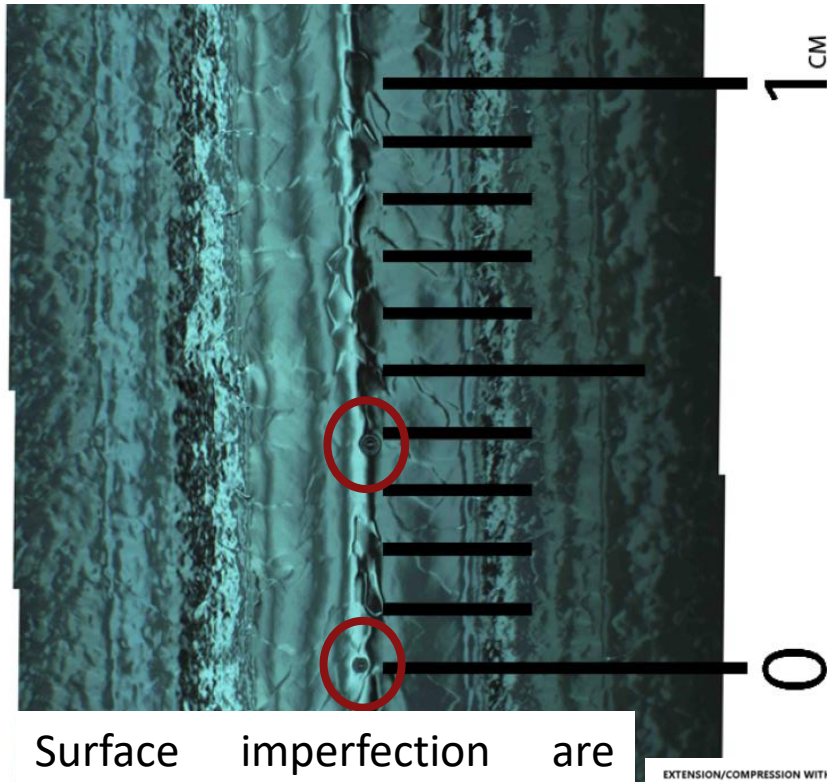
A. Godeke, Ph.D. dissertation

Nb₃Sn

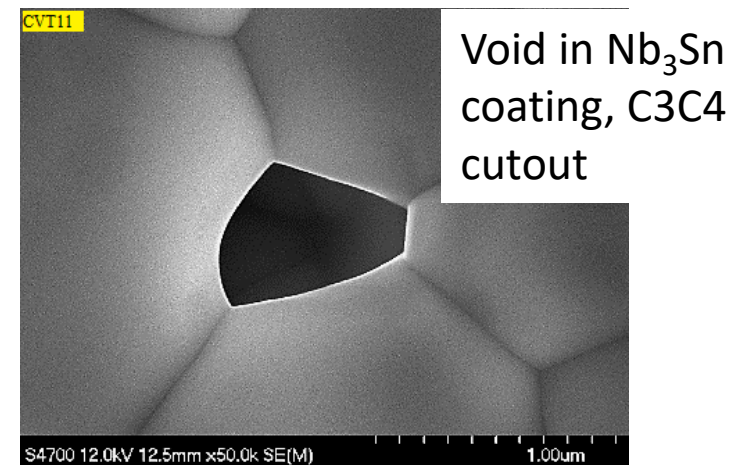
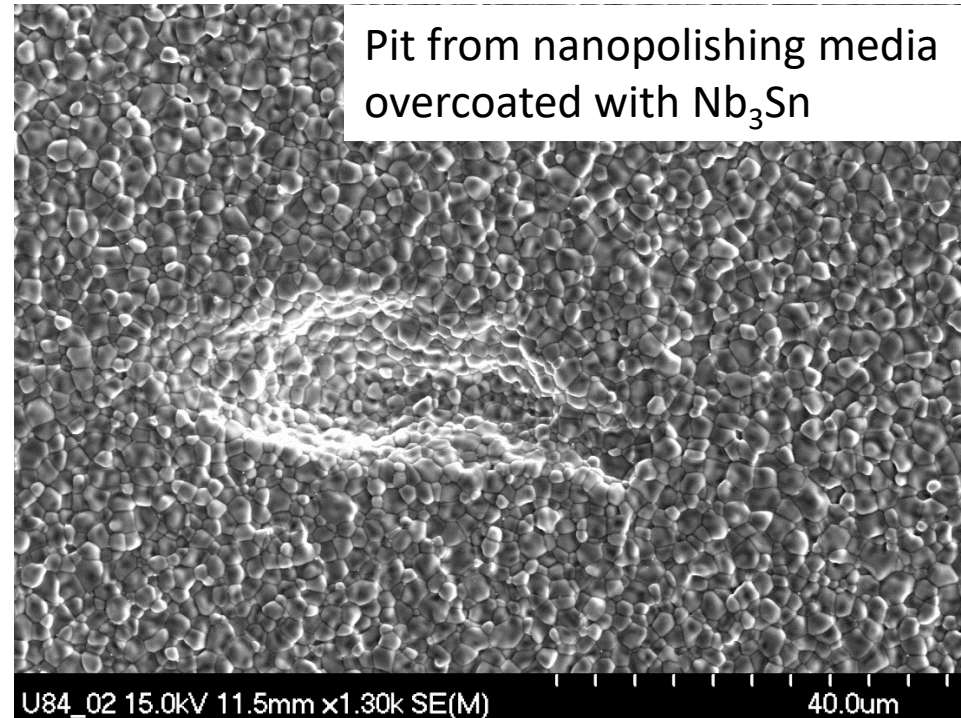
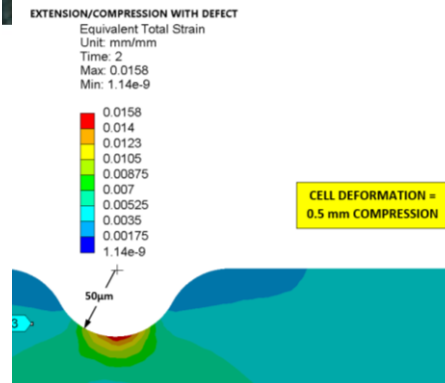


Expected degradation due to T_c reduction is not consistent with the measurement

Weak points?



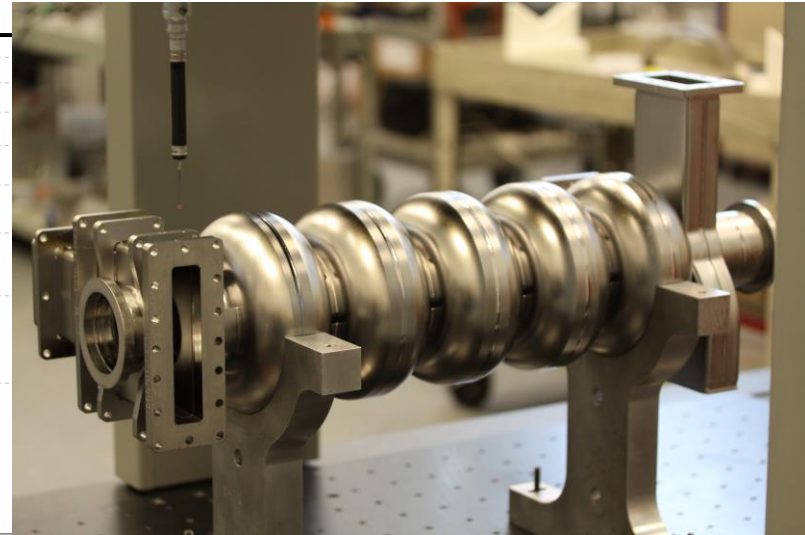
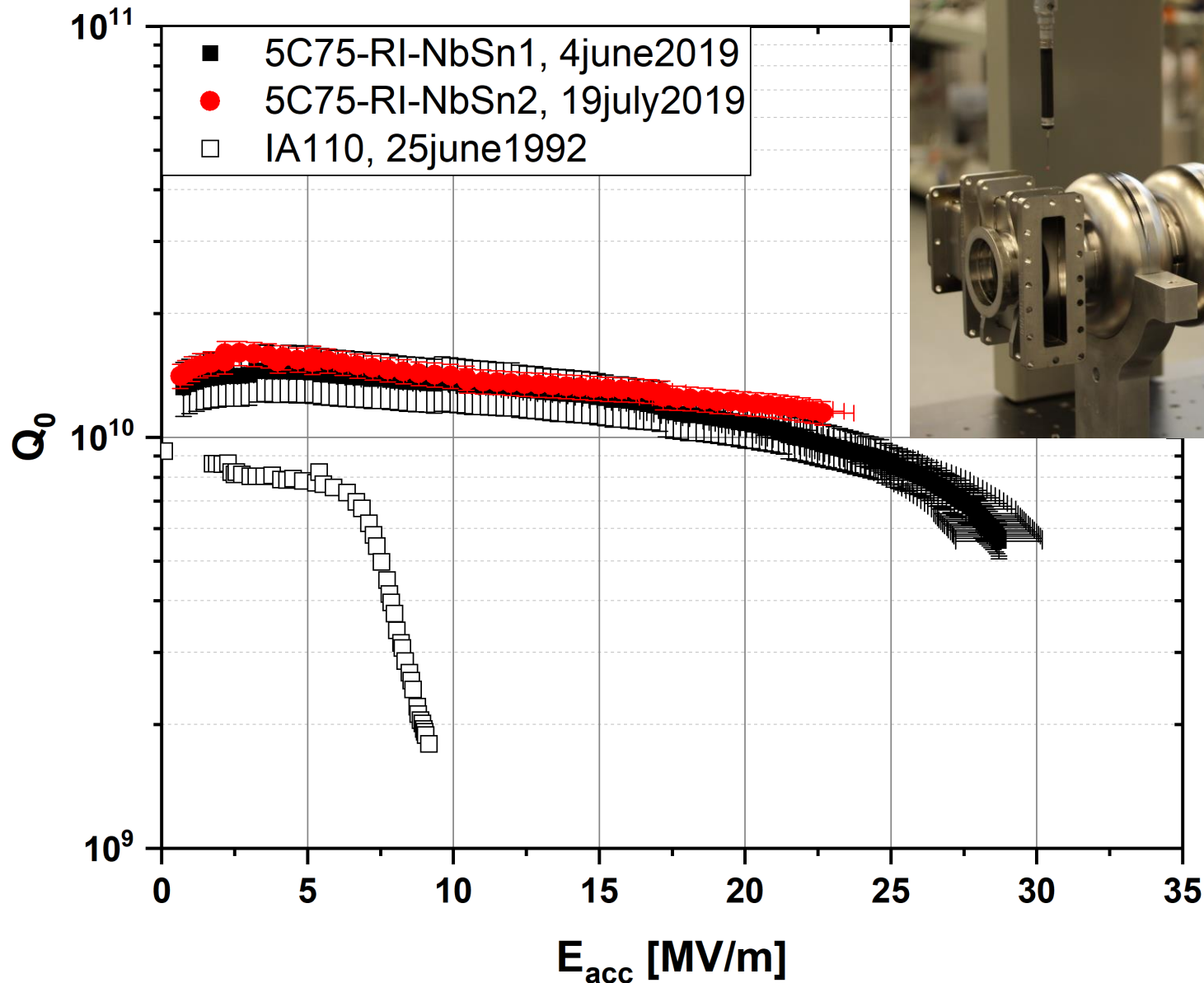
Surface imperfection are likely high stress points, where strain exceeds the average levels and significantly degrades surface resistance → smoothen the surface by centrifugal barrel polish



The quality of the substrate could be important...



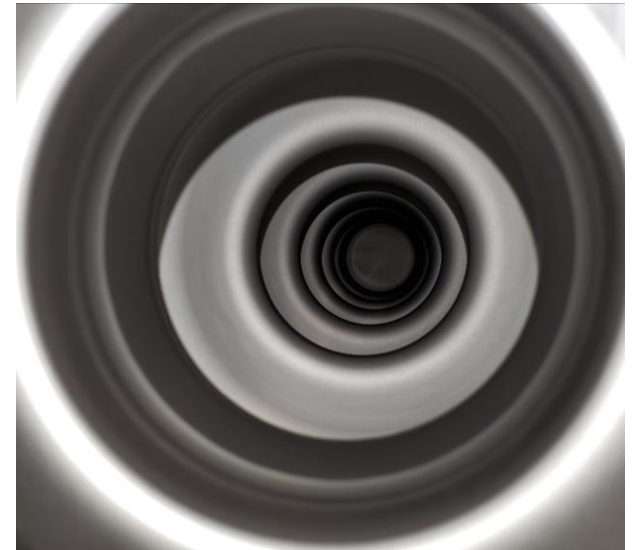
Baseline test of the new C75 cavities for Nb₃Sn project



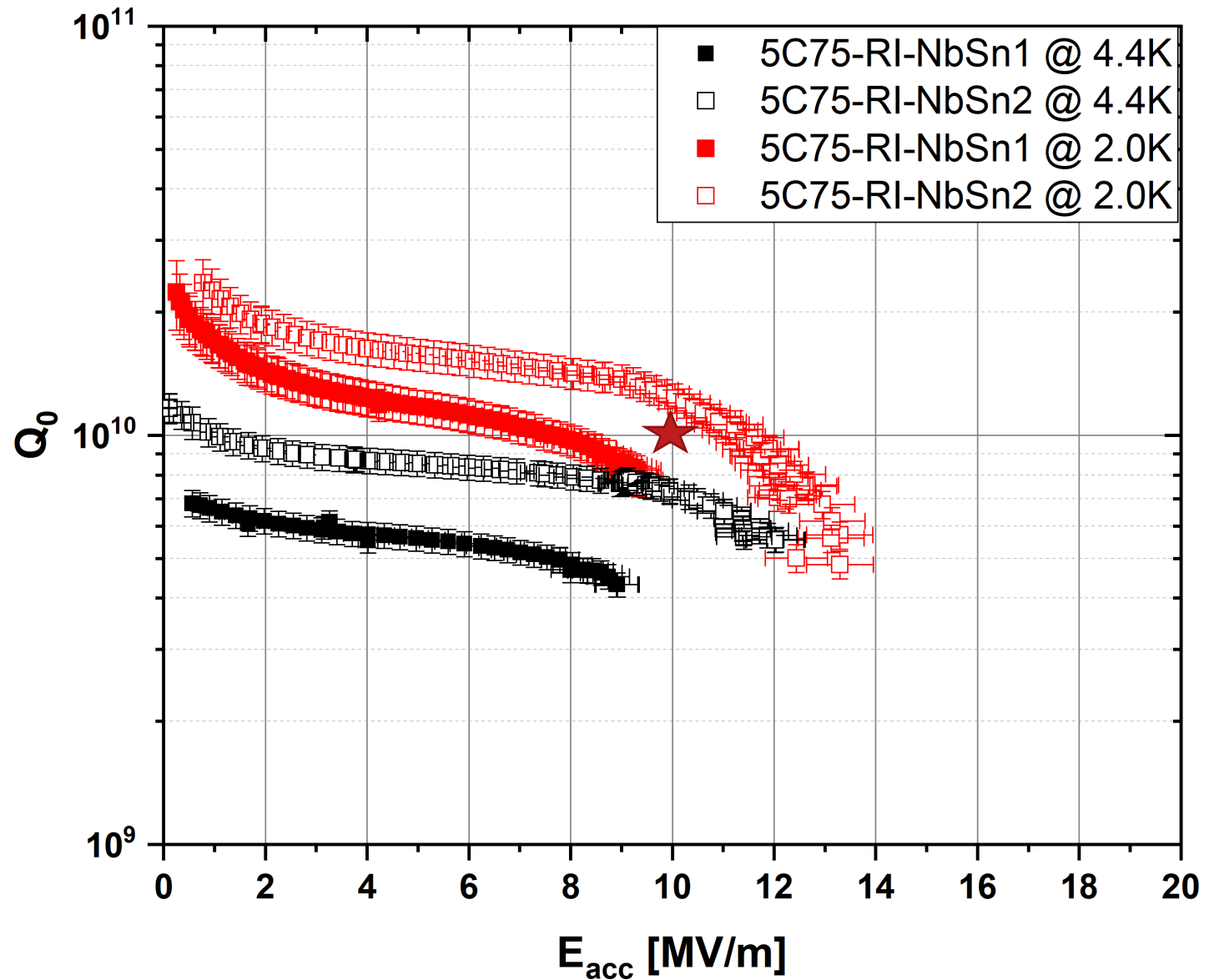
C75 cavity coating



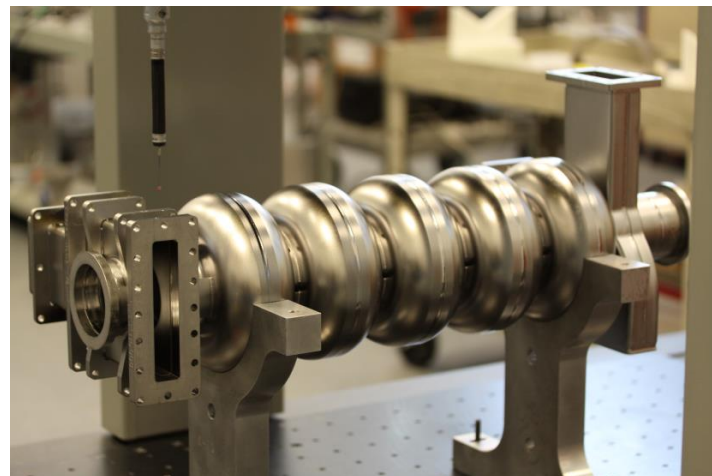
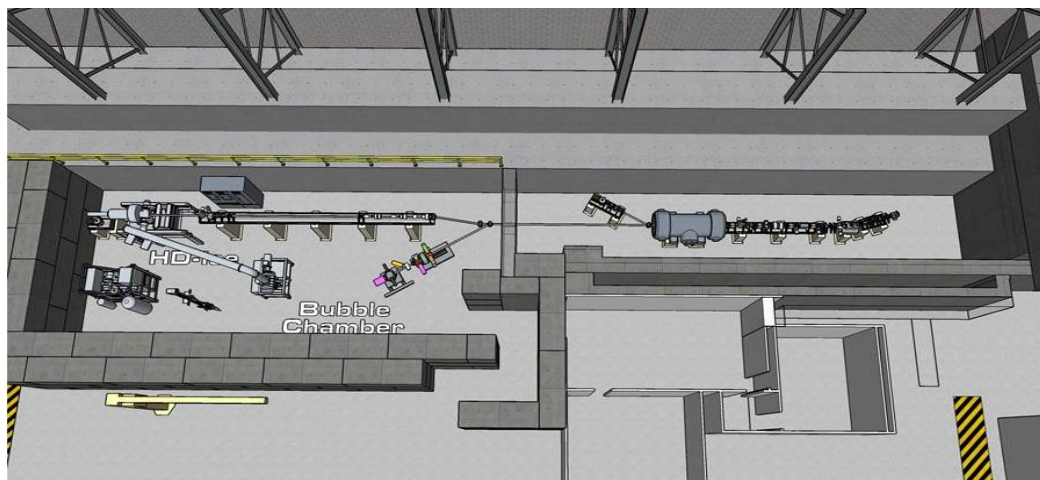
Uniform coating, no obvious asymmetry... in the second cavity!



Nb₃Sn-coated C75 cavity test results



Installation of Nb₃Sn-coated C75 into UITF



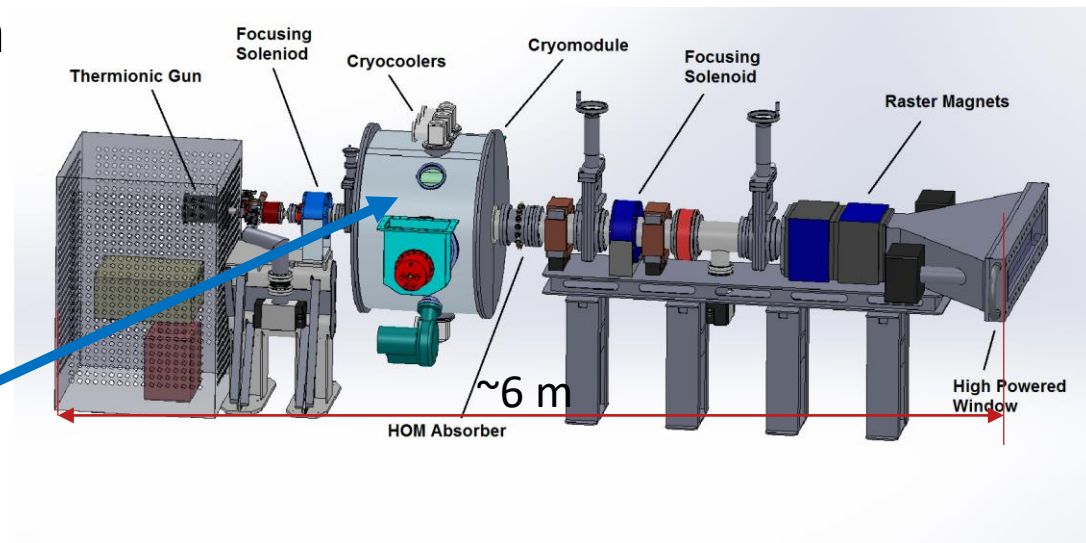
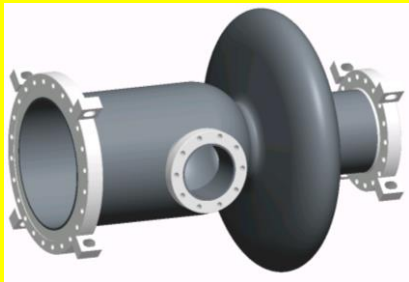
Application	Beam Energy	Beam Current	Experiment Duration	Notes	Presenter
Commission QCM for CEBAF	6 MeV, but prefer up to 10 MeV	up to 100 uA	three or four 1-week long tests	tests complete before long shutdown of 2020, when QCM to be installed at CEBAF	R. Kazimi
Commission HDice for CEBAF	~ 8 MeV	up to 100 nA for tuning, 0.25 to 5 nA for production	four or five run periods, one-month long each	target provides transverse polarization required for 3 A-rated Hall B experiments	A. Sandorfi
Manufacturing polarized targets for CEBAF via DNP	1 - 10 MeV	1 to 10 uA	hours, days	likely some R&D to determine optimum polarizing conditions	C. Keith
Bubble Chamber astrophysics	4 - 10 MeV	0.01 to 100 uA	3 weeks, as often as possible	UITF better location than CEBAF injector, when CEBAF shutdowns are short	R. Suleiman
MeV parity violation experiment	10 MeV	milliamps preferred, will reduce experiment duration	months to years	requires polarized electron beam, transmission geometry offers advantages	R. Carlini
Testing Nb ₃ Sn-coated cavities	determining the beam energy of test cavity is point of test	up to 100 uA	as many tests as possible	Nb ₃ Sn cavities require only 4K Helium	G. Ereemeev
Wastewater treatment	2- 10 MeV	100 uA	imagine week-long test durations over three years	together with local partners	G. Ciovati
Polarized positron source	5 - 10 MeV	up to 100 uA	staged tests, likely many required, 1-week long duration	requires polarized electron beam	J. Grames
EIC: fast kicker tests	5 - 10 MeV	up to 100 uA	two 1-week long tests	together with sbir-partner	H. Wang
EIC: testing high bunch charge	5 - 10 MeV	up to 100 uA	two 1-week long tests	requires polarized electron beam	J. Grames and J. Guo

Strain sensitivity is not necessarily an issue for new designs

Compact high-power CW SRF accelerator for industrial application

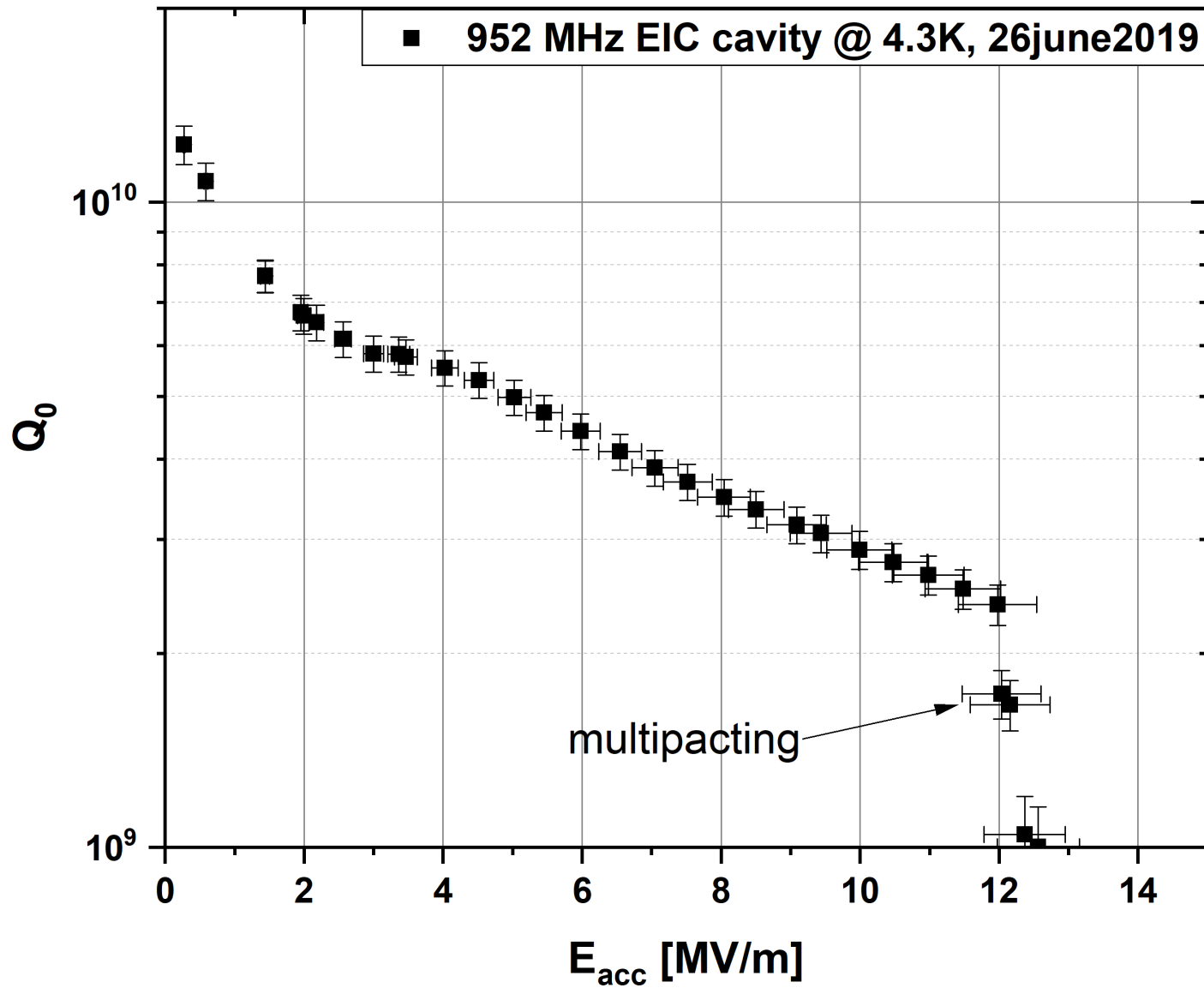
- 1-year design collaboration among JLAB, AES, General Atomics
- Funded by DOE-HEP (Accelerator Stewardship)
- Use in wastewater and flue-gas treatment
- 1 MeV, 1 A electron beam

$\text{Nb}_3\text{Sn}/\text{Nb}/\text{Cu}$ $\beta=0.5$ single-cell cavity, conduction cooled with four 1.5 W cryocoolers

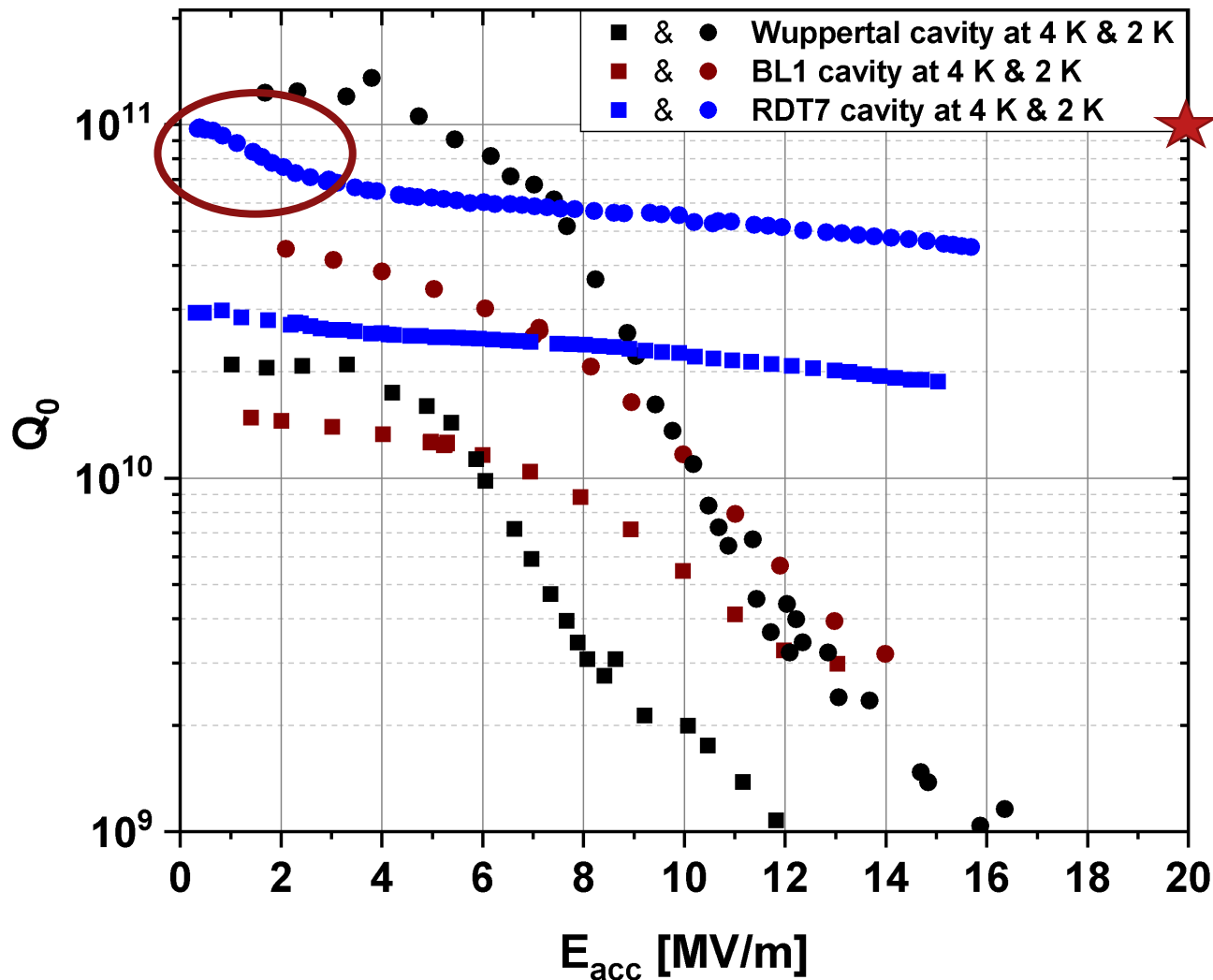


- Patent on Cryomodule design filed on 01/29/18

Nb₃Sn-coated 952 MHz cavity



Summary #1 : high-Q Nb₃Sn layers

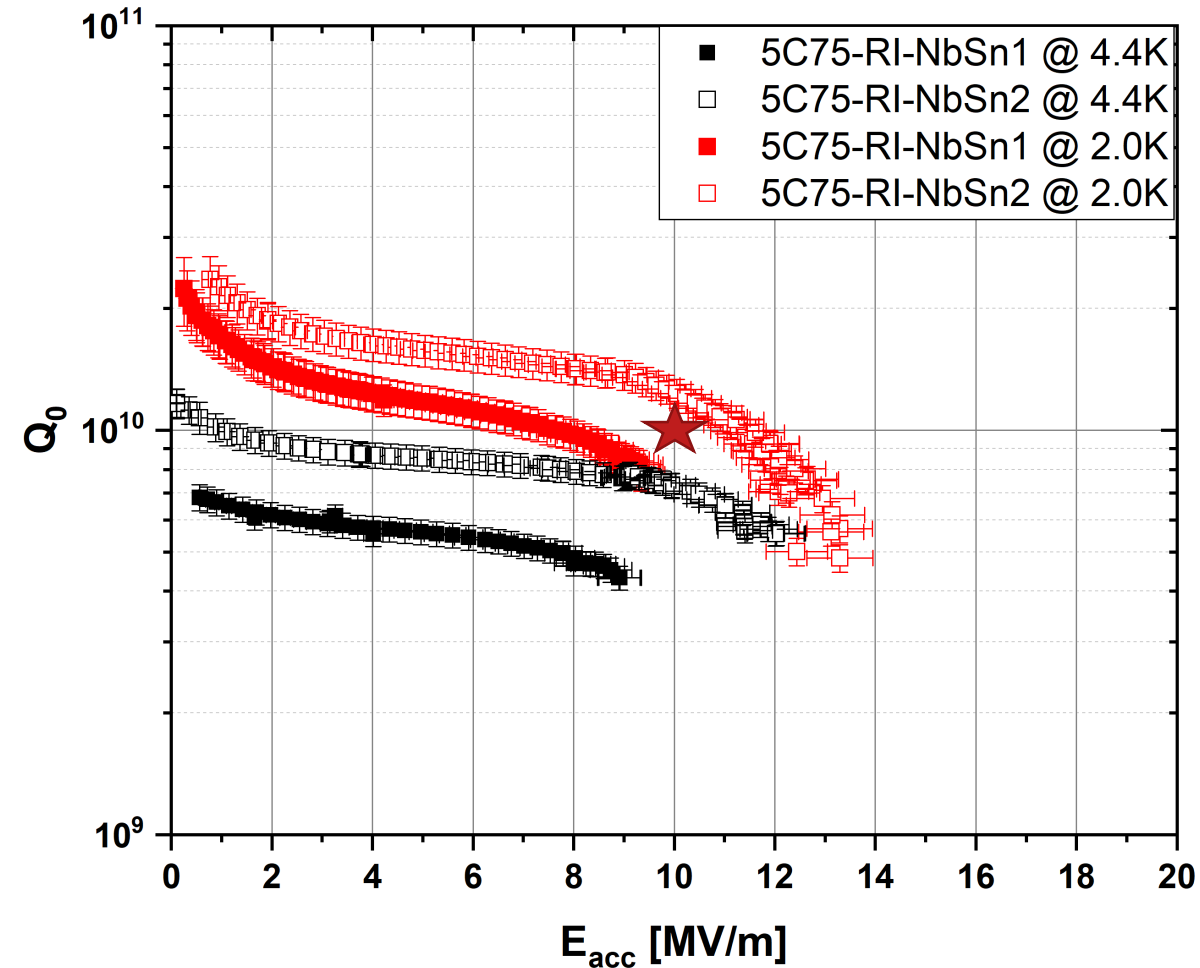


The goal is to optimize the coating process towards Q_0 of 10^{11} at $E_{acc} = 20$ MV/m at 2 K.

- Cavities w/o Q-slope were produced in “Siemens” configuration
- Q_0 of 10^{11} are measured at low fields
- Current focus is on low-field and medium field Q-slopes
- Temperature-controlled Sn source is being built
- It may be challenging to consistently reach $E_{acc} = 20$ MV/m w/o cleanroom around the coating system.

U. Pudasaini

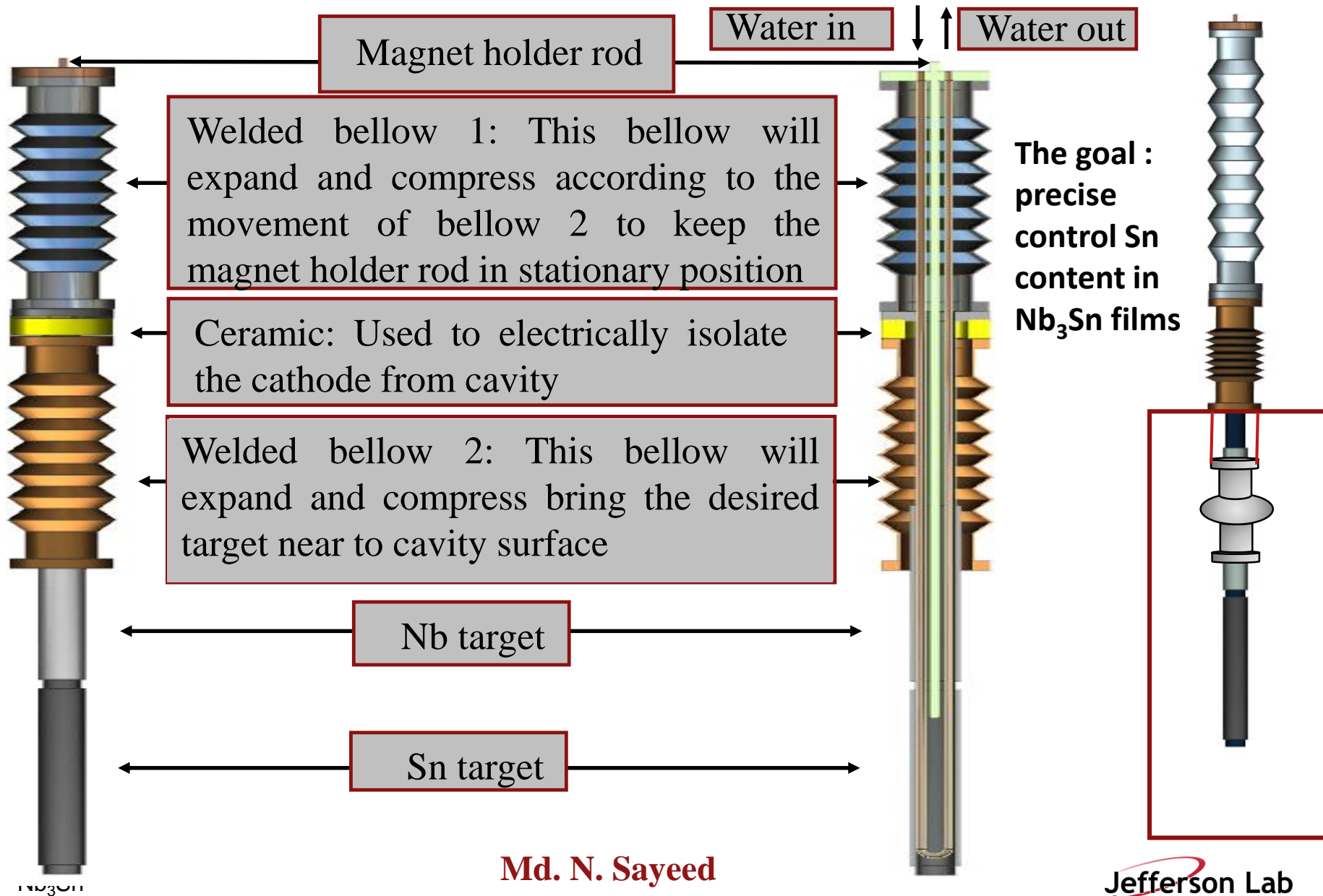
Summary #2 : Nb₃Sn for practical applications



The goal is to study coating degradation by accelerating electron beams in a cryomodule with Q_0 of 10^{10} at $E_{acc} = 10$ MV/m at 4 K.

- Substrate issues were resolved with the two new C75 cavities
- Discovered significant degradation after tuning likely related to surface features
- Possible mitigations are smooth surface and minimized tuning
- The best solution may involve redesign of a quarter cryomodule

Summary #3 : optimum Nb₃Sn layers



Thank you for your attention!